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Surface hardness measurement of NiP-plated AA7050

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Abstract

This paper is concerned with the surface hardness measurement of NiP-coated AA7050 using different loads from 10 to 100 g. The surface hardness was observed to increase from 180 to 600 H_v with increasing NiP layer thickness, depending on the load applied for indentation. When NiP coating thickness is thinner than 2 μm , the surface hardness of NiP-coated AA7050 was mainly determined by AA7050 substrate, while it was significantly increased by NiP coating layer when NiP coating thickness is thicker than 2 μm . Hardness of AA7050 substrate itself was not dependent on the applied load but the hardness of NiP-coated AA7050 was largely influenced by the load applied for indentation. The largest difference of hardness between 10 g and 100g of applied loads, was obtained at the NiP thickness of about 8 μm above which the measured hardness at 10 g reached a maximum value of about 600 H_v . It was also observed that indentation-induced plastic deformation next to the indented zone occurs when NiP layer is 5.64 times thicker than the depth of impression formed by indentation.

Keywords : Surface hardness, NiP, AA7050

1. Introduction

Aluminum 7050 alloy is a heat treatable alloy which has high toughness, high strength and high stress corrosion cracking resistance. It is commonly used for aeronautical structural components, extrusion parts and stamp forgings. However, its relatively low hardness together with high friction coefficient limits their further applications for tribological purpose in mechanical parts.

To improve the hardness and tribological performance of aluminum alloys, their surfaces need to be coated with hard anodizing films, plasma electrolytic oxidation coatings or hard metals. Anodized aluminum alloys showed increased hardness by decreasing electrolyte temperature [1, 2]. Hardness of anodic oxide films was increased by sealing treatments of the anodic oxide film in Ni-acetate solution, boiling water and NaAlO_2 solution [3]. Polytetrafluoroethylene (PTFE) and anodic aluminum oxide (AAO) composite film was fabricated by depositing PTFE particles into porous anodic aluminum oxide film using electrophoretic deposition (EPD) process [4]. Self-lubricating

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polytetrafluoroethylene-containing PEO (PEO-PTFE) composite coatings were prepared by vacuum impregnation on plasma electrolytic oxidized Al2024 alloy [5]. Improved tribological response of AA 2024 was achieved by DLC/NiP duplex coating [6]. Laser surface alloying of an A356 aluminum alloy using nickel and Ni-Ti-C powders improved its tribological behavior by providing a reinforcement of Al-Ni or Al-Si-Ti intermetallics and TiC particles [7].

Micro-Vickers hardness of aluminum alloys coated with anodic oxides was measured readily from the anodic oxide surface using an ink-impregnation method [8, 9]. Hardness of anodic oxide films is known to depend on the flaws and imperfections like vacant spaces within them [1, 9]. Measurement of micro-Vickers hardness was performed on NiP deposited on EN8 steel substrate at applied load of 500 g [10]. Hardness of electroless NiP deposited on aluminum alloys could be influenced not only by substrate but also by the load applied for hardness measurement if the NiP coating thickness is not thick enough. At present, the effects of NiP coating thickness on the hardness measured from the NiP coating surface on an aluminum alloy has not been fully understood.

In this paper, the effects of NiP coating thickness and applied load for micro-Vickers hardness measurement on the NiP coating surface on AA7050 were investigated and plastic deformation behavior of NiP layer and substrate was discussed in view of the ratio of NiP layer thickness to depth of impression formed by indentation.

2. Experimental

Al7050 alloy disc samples (wt.%, Cu 2.3, Mg 2.3, Zn 6.2, Zr 0.12, Fe and Si < 0.1 and Al balance) with diameter of 25mm and thickness of 2 mm, were abraded by SiC papers successively up

to # 2000 SiC paper, washed with tap water, dried with an air stream and then used for experiments. The disc samples were first immersed in a commercial zincating solution for 2 min at RT. The first zincate-treated samples are etched in nitric acid solution and then immersed again in the same zincating solution for 2 min to form a zincate layer of good adhesion. The double zincate-treated samples were immersed to form NiP layers in a commercial electroless NiP plating solution for various durations up to 120 min at 80 °C. The zincating pre-treatment is employed to improve the adhesion of the NiP layer prepared by electroless plating method. Thickness of the Ni-P layer was obtained from the cross-sectional observation using SEM (Scanning Electron Microscopy, JSM-6610LV). Micro-Vickers hardness was measured on the NiP layer surface using micro-Vickers hardness tester (V-test II, Bareiss) with a pyramidal-diamond indenter at four different loads of 10, 20, 50 and 100 g. The hardness was measured at five points on each specimen and their average value was obtained. The impressions remaining after indentation were observed using optical microscope (3D digital microscope, HiROX) and SEM.

3. Results and discussion

Cross-sections of NiP-coated AA7050 and their thicknesses are presented in Fig. 1 as a function of NiP electroplating time. The NiP layer was formed uniformly over the entire surface of AA7050 by electroless plating after double zincate treatment. The NiP coating layer thickness increased linearly with plating time at a growth rate of about 0.17 $\mu\text{m}/\text{min}$.

Micro-Vickers hardness of NiP-coated AA7050 was measured from the NiP surface and the surface hardness values are plotted against electroless plating time and NiP

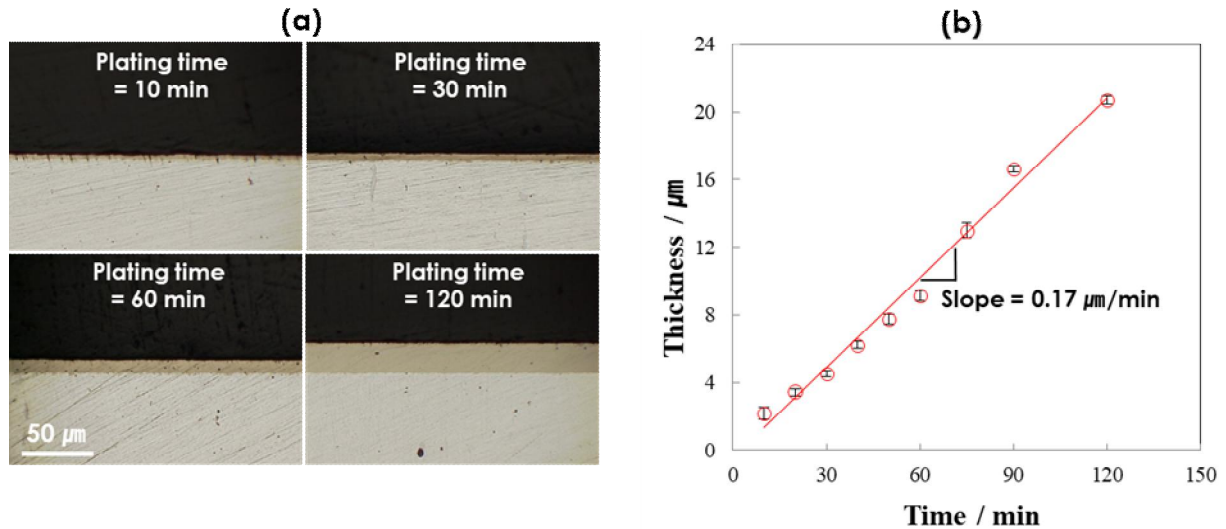


Fig. 1. (a) Cross-sections and (b) thickness of Ni-P layer formed on AA7050 with electroless plating time

thickness in Fig. 2(a) and 2(b), respectively. Hardness of the NiP-coated AA7050 surface increased with increasing NiP plating time and NiP thickness, suggesting improved surface hardness of AA7050 by NiP coating. The surface hardness of NiP-coated AA7050 was determined mainly by AA7050 substrate, if NiP coating thickness is thinner than 2 μm. On the other hand, the surface hardness is largely influenced by NiP layer if NiP coating thickness is thicker than 2 μm. It is interesting to note that the surface hardness of NiP-coated AA7050 reaches a maximum value of about 600 Hv, if NiP coating layer thickness is thicker than a critical value of

about 8 μm.

In Fig. 2(b), it should be pointed out that the hardness of NiP-coated AA7050 depends on not only the NiP coating thickness but also the load applied for indentation. Figure 3 shows changes in the surface hardness of NiP-coated AA7050 with the applied load for indentation. It is apparent that hardness of AA7050 substrate itself does not depend on the applied load but the hardness of NiP-coated AA7050 is significantly decreased with increasing applied load for indentation.

In order to understand why the surface hardness of NiP-coated AA7050 decreases with the applied load for indentation, it is

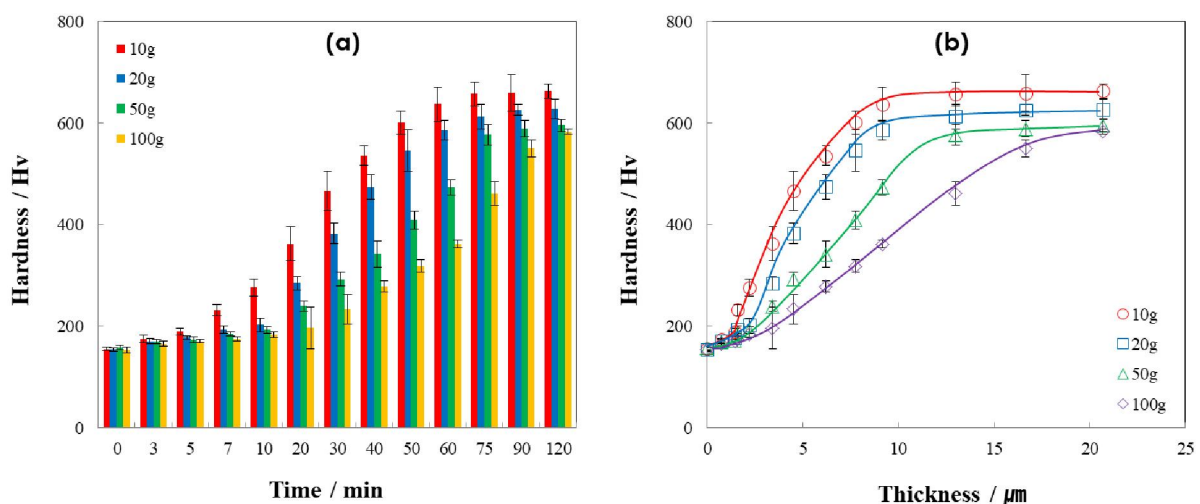


Fig. 2. Hardness of NiP-plated Al7050 alloy with (a) electroless plating time and (b) NiP layer thickness. The hardness was measured at four different loads of 10, 20, 50 and 100 g.

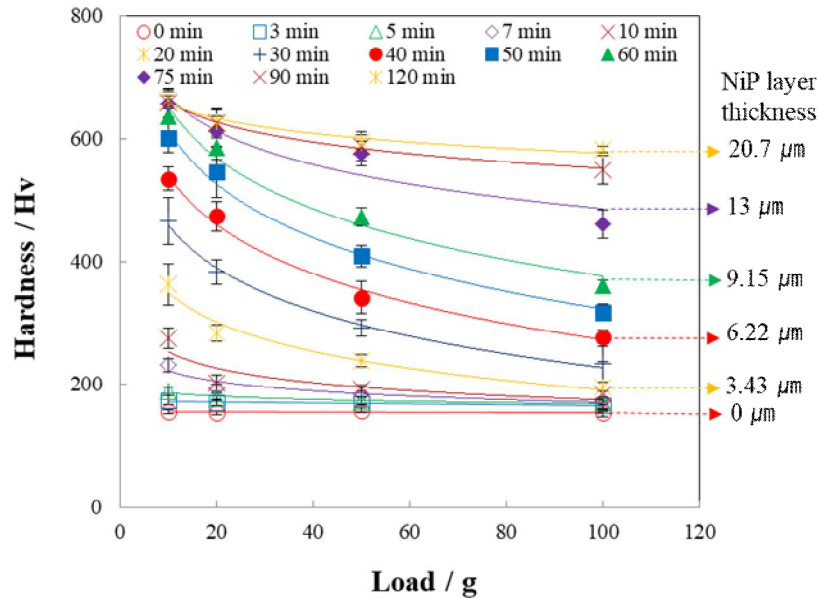


Fig. 3. Hardness of NiP-plated Al7050 alloy with applied load for indentation.

necessary to assume that not only NiP layer but also AA7050 alloy substrate are deformed when it is indented from the NiP layer surface. Since more plastic deformation occurs by applying higher load for indentation, AA7050 substrate could contribute more significantly to the measured hardness with increasing the load for indentation. This could explain the reduced hardness of NiP-coated AA7050 with applied load in Fig. 3.

The magnitude of decreased surface hardness on NiP-coated AA7050 samples by changing the applied load from 10 g to 100g, is plotted against NiP layer thickness in Fig. 4. The largest reduction of hardness between 10 g and 100g of applied loads was obtained at the NiP thickness of about 8 μm above which the measured hardness at 10 g reached a steady-state value of about 600 H_v . This suggests that the largest difference of hardness between 10 g and 100g of applied loads is obtained when most of plastic deformation by indentation using 10 g of load occurs within the NiP layer and the impression at 100 g of load is formed by plastic deformation of not only NiP layer but also AA7050 substrate.

In general, more than 600 H_v of surface hardness of NiP-plated AA7050 is required

for practical use in contact with sliding balls for an intermediate shaft in cars. To use AA7050 for the intermediate shaft, its surface should become hard enough not to be indented by sliding balls. Assuming that the surface hardness of about 600 H_v at the load of 100 g is sufficient for use with sliding balls in the intermediate shaft, it can be said that NiP layer should be thicker than 20 μm . The minimum thickness of NiP layer not to be indented by the sliding balls in an intermediate shaft is to be examined further.

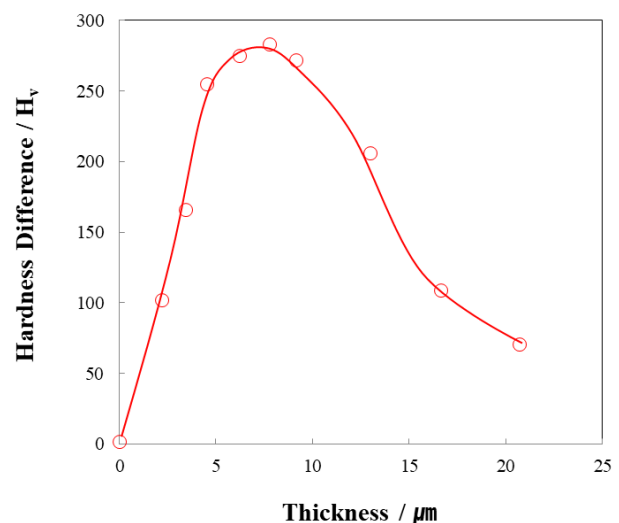


Fig. 4. Hardness difference of NiP-plated Al7050 alloy between the highest and lowest values measured at different loads in Fig. 3.

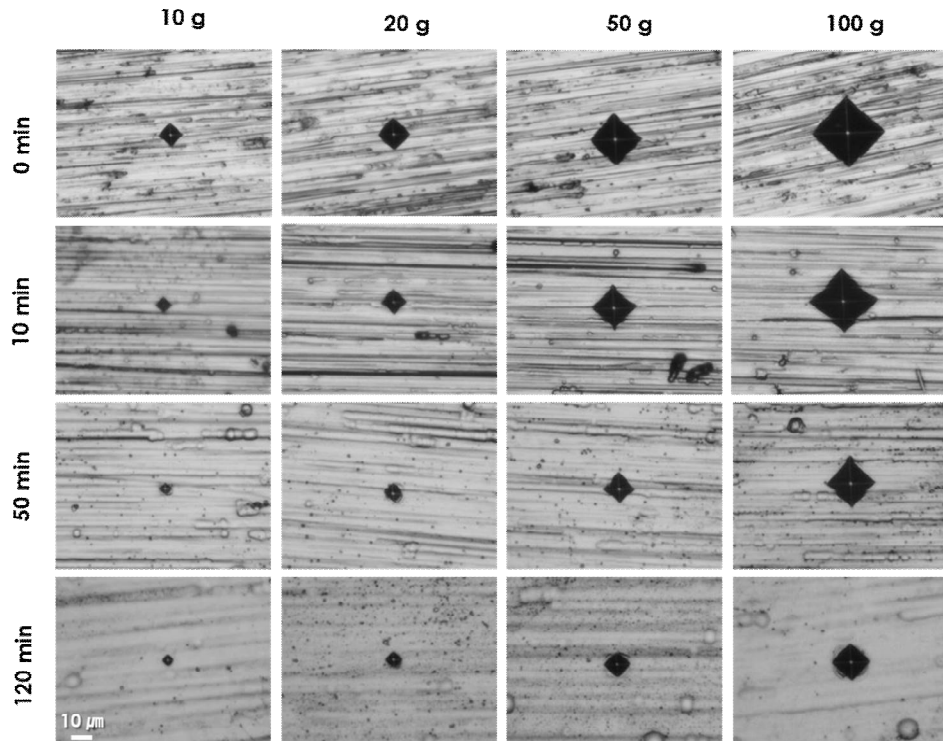


Fig. 5. Optical microscope images of Al7050 alloy plated with Ni-P for different durations from 0 to 120 min after indentation at various applied loads from 10 to 100 g.

Size of the indentation formed by pyramidal indenter increased with increasing load and it decreased with increasing NiP plating time, as exhibited in Fig. 5. The reduced indentation size with NiP plating time reveals that the surface of NiP-plated AA7050 becomes more resistive against an external force as the NiP layer thickness increases.

The shapes of indentation impressions formed by pyramidal indenter are examined in more detail by SEM and the results are displayed in Fig. 6. The indented surfaces of AA7050 substrate without NiP coating and 3.5 μm thick NiP-coated specimen showed a dark single cross line at the center of the impression but 9.6 and 21.3 μm thick NiP-coated surfaces exhibited bright double bands along with a dark cross line at the center of the impression. Another interesting observation is the formation of indentation-induced plastic deformation next to the indented zone, as indicated by arrows in Fig. 6(d). On the other hand, the indentation-induced plastic deformation next to the indented zone was not observed when the NiP layer thickness of 9.6 μm is 1.83 times thicker than 5.23 μm of the depth of impression, as observed in Fig. 6(c).

By considering the angle of 136° of a pyramidal indenter used for micro-Vickers hardness measurement, depth of impression (h) formed by the indenter can be calculated using Eq. (1)

$$h = (D/2) \times \tan 22^\circ \quad (1)$$

where D is the average size of impression diagonals.

Table 1 displays h and $1/h$ for different NiP layer thicknesses from 0 to 21.3 μm . Depth of impression formed on the 21.3 μm thick NiP-coated AA7050 surface at 100 g of load (Fig. 6(d)) was calculated to be about 3.77 μm . Considering that the NiP layer thickness is 5.64 times thicker than the depth of impression, it can be said that plastic deformation of the 21.3 μm thick NiP-coated AA7050 occurs only within the NiP coating layer during indentation at 100g of load. This may explain why the indentation-induced plastic deformation next to the indented zone occurs only when the NiP coating thickness is much thicker than the depth of impression formed by a pyramidal indenter in Fig. 6(d). Further study on

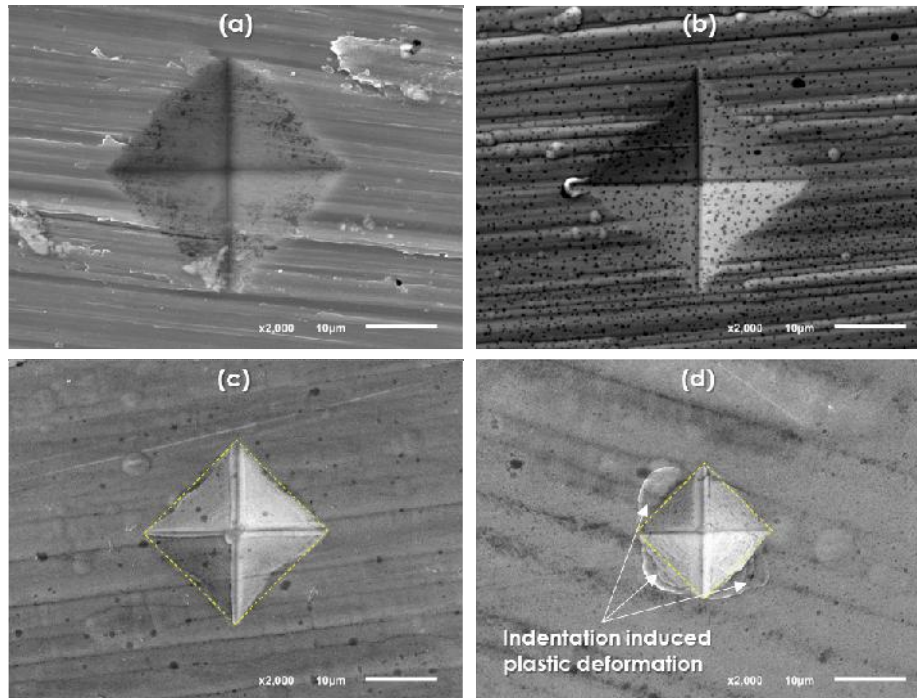


Fig. 6. SEM images of indented regions of AA7050 alloy with NiP coating thickness of (a) 0, (b) 3.5, (c) 9.6 and (d) 21.3 μm . The indentation was conducted at an applied load of 100 g.

Table. 1. NiP layer thickness (l), average size of impression diagonals (D), depth of impression (h) and ratio of l/h on AA7050. The applied load for indentation was 100 g.

| NiP layer thickness, l [μm] | 0 | 3.5 | 9.6 | 21.3 |
|---|-------|-------|-------|-------|
| Average size of impression diagonals, D [μm] | 33.04 | 33.04 | 25.91 | 18.70 |
| Depth of impression, h [μm] | 6.67 | 6.67 | 5.23 | 3.78 |
| l/h ratio | - | 0.52 | 1.83 | 5.64 |

the relationship between indentation-induced plastic deformation and thickness of NiP layer is necessary.

There were observed black spots on the NiP-coated AA70705 surface. The number of the black spots decreased while their size increased with increasing NiP coating thickness. The black spots seem to be formed by corrosion through pinhole defects within the NiP coating layer.

4. Conclusions

Hardness of NiP-coated AA7050 was measured from the NiP coating surface using different loads from 10 to 100 g and it was analyzed as a function of the NiP layer

thickness and applied load for indentation. The following results are obtained in this work.

1. The surface hardness of AA7050 was significantly improved from 180 to 600 H_v by coating with NiP layer.

2. The surface hardness of NiP-coated AA7050 was mainly determined by AA7050 substrate when NiP coating thickness is thinner than 2 μm . On the other hand, the surface hardness is largely influenced by NiP layer when NiP coating thickness is thicker than 2 μm .

3. Hardness of AA7050 substrate itself does not depend on the applied load but the hardness of NiP-coated AA7050 is significantly decreased with increasing applied load for indentation.

4. The largest difference of between 10 g

and 100g of applied loads was obtained at the NiP thickness of about 8 μm when the measured hardness at 10 g reached a maximum value of about 600 H_v . The largest difference of hardness between 10 g and 100g of applied loads is explained by the facts that most of plastic deformation at 10 g of load occurs within the NiP layer while large portion of plastic deformation occurs within AA7050 substrate at 100 g of load.

5. Indentation-induced plastic deformation next to the indented zone was observed only when NiP layer is thick enough more than 5 times comparing with depth of impression formed by indenter.

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