



## Cu<sub>2</sub>O Thin Film Photoelectrode Embedded with CuO Nanorods for Photoelectrochemical Water Oxidation

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### Abstract

Assembling heterostructures by combining dissimilar oxide semiconductors is a promising approach to enhance charge separation and transfer in photoelectrochemical (PEC) water splitting. In this work, the CuO nanorods array/Cu<sub>2</sub>O thin film bilayered heterostructure was successfully fabricated by a facile method that involved a direct electrodeposition of the Cu<sub>2</sub>O thin film onto the vertically oriented CuO nanorods array to serve as the photoelectrode for the PEC water oxidation. The resulting copper-oxide-based heterostructure photoelectrode exhibited an enhanced PEC performance compared to common copper-oxide-based photoelectrodes, indicating good charge separation and transfer efficiency due to the band structure realignment at the interface. The photocurrent density and the optimal photocurrent conversion efficiency obtained on the CuO nanorods/Cu<sub>2</sub>O thin film heterostructure were 0.59 mA/cm<sup>2</sup> and 1.10% at 1.06 V vs. RHE, respectively. These results provide a promising route to fabricating earth-abundant copper-oxide-based photoelectrode for visible-light-driven hydrogen generation using a facile, low-cost, and scalable approach of combining electrodeposition and hydrothermal synthesis.

*Keywords:* Copper oxide, Oxide heterostructure, photoelectrode, Photoelectrochemical water oxidation, Hydrogen production

## 1. Introduction

Harvesting sunlight to produce clean chemical fuels such as hydrogen is considered to be a desirable and sustainable method toward meeting future demands for energy with minimal environmental impact [1-3]. The development of advanced processes to generate hydrogen from the solar-driven splitting of water has been attempted through several different approaches. Among these methods [4-6], photoelectrochemical (PEC) water splitting is known to be one of the most

promising hydrogen production techniques in an eco-friendly and highly-efficient way. In general, solar-driven water splitting involves two half-cell reactions, (i) the oxygen evolution reaction or water oxidation and (ii) the hydrogen evolution reaction or water reduction, and thus it is necessary to separate each half-cell reaction [3]. One of the main reasons why PEC cells are widely used as solar-driven water splitting devices is that they spatially separate water oxidation and reduction reactions.

In a PEC cell system, the selection and design of the photocatalytic materials for the water oxidation and reduction photoelectrodes are critical, because the light absorption and carrier transport of the photoelectrodes largely determine the capability of the PEC cell for water splitting

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[7]. It has been shown that oxide semiconductors such as titanium dioxide ( $\text{TiO}_2$ ), hematite ( $\text{Fe}_2\text{O}_3$ ), zinc oxide ( $\text{ZnO}$ ), cuprous oxide ( $\text{Cu}_2\text{O}$ ), and cupric oxide ( $\text{CuO}$ ) exhibit a promising potential as photoelectrodes for hydrogen generation [8-10]. Among them  $\text{Cu}_2\text{O}$  and  $\text{CuO}$ , which have a direct bandgap of approximately 2.0 eV and 1.5 eV respectively [3], are attractive in terms of their great potential of earth-abundant, visible-light activated photoelectrodes for PEC water splitting owing to their direct bandgaps of small energies. However, both unmodified  $\text{Cu}_2\text{O}$  and  $\text{CuO}$  have significant electron-hole recombination rate and poor water splitting kinetics. To address these restrictions, various strategies including nanostructured control and elemental doping have been attempted [11]. In principle, constructing heterostructure is the most effective and direct way to promote the efficiency of charge separation in photoelectrodes [6]. Since both the conduction band (CB) and valence band (VB) edges of  $\text{Cu}_2\text{O}$  are higher than those of  $\text{CuO}$  [12], the photoinduced electrons in the CB of  $\text{Cu}_2\text{O}$  are supposed to be injected into that of  $\text{CuO}$  at the  $\text{CuO}/\text{Cu}_2\text{O}$  interface under visible-light illumination, while the photoinduced holes in the VB of  $\text{CuO}$  are injected into that of  $\text{Cu}_2\text{O}$ . As a result, for example, the  $\text{CuO}/\text{Cu}_2\text{O}$  bilayered heterostructure has been shown to facilitate electron-hole separation and can improve photo-to-chemical energy conversion efficiency [3]. Meanwhile, as we have seen, p-type semiconductors  $\text{Cu}_2\text{O}$  and  $\text{CuO}$  in the form of individual layer have been used as photocathode for hydrogen evolution reaction or water reduction [9]. On the contrary, as described above, the engineered  $\text{CuO}/\text{Cu}_2\text{O}$  electronic structure at the interface is expected to provide a copper-oxide-based heterostructure as an effective photoanode for PEC oxygen evolution reaction or water oxidation, which is the main point of this work.

In this work, we assemble  $\text{Cu}_2\text{O}$  thin film photoelectrode embedded with  $\text{CuO}$  nanorods by directly electrodepositing  $\text{Cu}_2\text{O}$  thin film onto the vertically oriented  $\text{CuO}$  nanorods array. As illustrated in Figure 1, the step-wise structure of energy levels constructed  $\text{CuO}/\text{Cu}_2\text{O}$  heterostructure

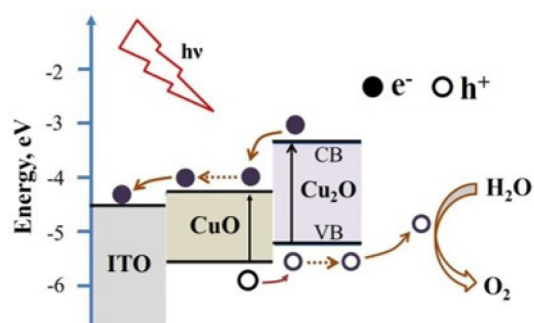


Fig. 1. Schematic diagram of the mechanism for the photoelectrochemical water oxidation at  $\text{CuO}/\text{Cu}_2\text{O}$  heterostructure photoanode.

is responsible for the mechanism of PEC photoanode. When  $\text{CuO}$  and  $\text{Cu}_2\text{O}$  come into contact, an oxide semiconductor heterostructure forms at their interface by coupling dissimilar semiconductors. Consequently, there is a favorable energy band alignment for electron transport from the CB of  $\text{Cu}_2\text{O}$  to the CB of  $\text{CuO}$ , and for hole transport from the VB of  $\text{CuO}$  to the VB of  $\text{Cu}_2\text{O}$ . Under the illumination, the photoinduced electrons and holes are supposed to be efficiently separated by the built-in space-charge-region potential due to the band structure realignment at the interface, which also lowering the electron-hole recombination rate [13]. On the one hand, the electrons transport through each individual  $\text{CuO}$  nanorod and finally reach the ITO electrode to export the PEC electrical signals. On the other hand, the holes migrate through  $\text{Cu}_2\text{O}$  film and subsequently take part in the water oxidation at the surface of  $\text{Cu}_2\text{O}$  [14]. The nearly one-dimensional nature of  $\text{CuO}$  nanorods array is supposed to induce a significant increase in the interfacial area between  $\text{CuO}$  and  $\text{Cu}_2\text{O}$ , and the introduction of  $\text{CuO}$  layer is expected to results in a marked improvement in the visible light absorption due to its narrower band gap. Simultaneously, the engineered  $\text{CuO}/\text{Cu}_2\text{O}$  heterostructure is presumed to ensure a faster carrier transportation rate inside the copper oxides, which benefits the PEC water oxidation. This work demonstrates the potential of the  $\text{Cu}_2\text{O}$  thin film embedded with  $\text{CuO}$  nanorods as a photoanodic material for effective PEC water oxidation.

## 2. Experimental Details

The vertically ordered CuO nanorod array was prepared on a glass substrate coated with an indium tin oxide (ITO) electrode via a seed mediated hydrothermal technique with the use of a CuO nanoparticle seed layer formed via a two-step method [15]. To form the CuO seed layer, Cu metal film was first deposited on the ITO-coated glass substrate from a pure Cu metal target. The Cu film was deposited under an Ar atmosphere of 5 mTorr for 6 min at room temperature (RT) by using a radiofrequency magnetron sputtering system with power of 5 W. Subsequently, the as-deposited Cu film was thermally oxidized in dry air at 400°C for 1 h by using a rapid thermal annealing system to form CuO nanoparticles, which acted as nucleation sites for CuO nanorods. Aligned CuO nanorods were prepared by dipping the CuO-deposited substrate into a solution which consisted of 25 mM  $\text{Cu}(\text{NO}_3)_2 \cdot 3\text{H}_2\text{O}$  and 25 mM  $\text{C}_6\text{H}_{12}\text{N}_4$  in distilled water, and then heating it at 80°C for 2 h in an oven. Before the formation of an oxide heterostructure, the substrate covered with CuO nanorods was carefully cleaned with deionized (DI) water and dried under a high-purity nitrogen gas flow.

Next, we fabricated  $\text{Cu}_2\text{O}$  thin film embedded with CuO nanorods by directly electrodepositing  $\text{Cu}_2\text{O}$  thin film onto the CuO nanorods array. The electrodeposition of  $\text{Cu}_2\text{O}$  thin film was conducted with a three-electrode system in water bath of 60°C [14]. Aqueous solution containing 0.05 M  $\text{CuSO}_4 \cdot 5\text{H}_2\text{O}$  and 0.1 M lactic acid was prepared, and the pH value was adjusted to 11 through

dropwise adding NaOH. Afterward, the substrate covered with CuO nanorods array was immersed into the resulting solution under the current of 0.5 mA/cm<sup>2</sup> for 30 min. We used a calomel [ $\text{Hg}/\text{Hg}_2\text{Cl}_2/\text{KCl}$  (3.5M)] as the reference electrode and a Pt sheet as the counter electrode. Finally, the fabricated CuO/ $\text{Cu}_2\text{O}$  heterostructure was again rinsed with DI water for three times and dried in dry air.

The crystalline phases of the prepared CuO nanorods/ $\text{Cu}_2\text{O}$  thin film heterostructure were characterized by X-ray diffraction (XRD) and Raman spectroscopy, and then its microstructures were analyzed by scanning electron microscopy (SEM). All the PEC measurements for the CuO/ $\text{Cu}_2\text{O}$  photoanode were performed in a three-electrode cell with our photoanode as the working electrode, a Pt sheet as counter electrode, and a calomel reference electrode with an aqueous 0.1 M  $\text{Na}_2\text{SO}_4$  solution pH-adjusted to 5.8 as the electrolyte. The area of the photoanode exposed to light was 0.5 cm<sup>2</sup>. The photoresponse was measured under a continuous irradiation from a 300 W Xe lamp, and the visible light intensity were calibrated to 1 mW/cm<sup>2</sup> for the incidence onto the photoanode.

## 3. Results and Discussion

Typical SEM images for the top-view morphology of the hydrothermally synthesized CuO nanorods array and electrodeposited  $\text{Cu}_2\text{O}$  thin film on the ITO-coated glass substrate are presented in Figure 2(a) and 2(b), respectively. It is clearly seen from Figure 1(a) that the CuO nanorods having an average diameter of ~30 nm

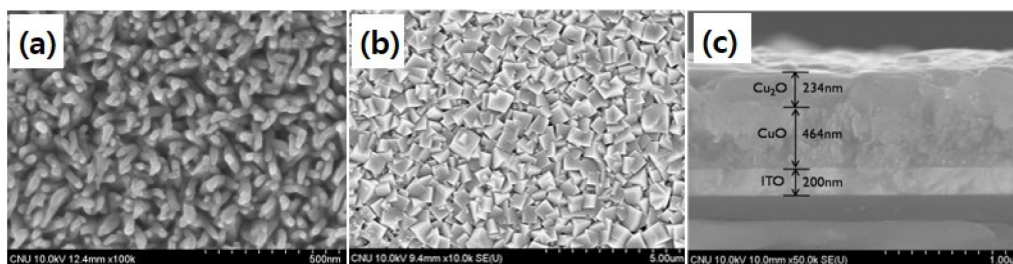


Fig. 2. Typical SEM images showing (a) top view of hydrothermally synthesized CuO nanorods array, (b) top view of electrodeposited  $\text{Cu}_2\text{O}$  thin film at pH 11, and (c) cross-sectional view of CuO nanorods/ $\text{Cu}_2\text{O}$  thin film heterostructure fabricated on an ITO-coated glass substrate.

are vertically grown. As shown in Figure 2(b), the electrodeposited  $\text{Cu}_2\text{O}$  thin film from lactate-stabilized copper sulfate solution of pH 11 is composed of closely packed  $\text{Cu}_2\text{O}$  nanocrystals that have an average grain size of  $\sim 500$  nm. The average grain size of  $\text{Cu}_2\text{O}$  was found to be easily adjusted by controlling the pH value of the electrolyte solution, but the most uniform and compact  $\text{Cu}_2\text{O}$  thin film could be obtained at pH = 11.

Figure 2(c) shows a typical SEM image for the cross-sectional morphology of the CuO nanorods/ $\text{Cu}_2\text{O}$  thin film heterostructure fabricated on the ITO-coated glass substrate. As clearly seen in this image, the surface of the distinct ITO layer is covered by vertically aligned CuO nanorods with a length of  $\sim 450$  nm. Also, the  $\text{Cu}_2\text{O}$  thin film with a thickness of  $\sim 250$  nm is found to cover all over the CuO nanorods array. It is worth pointing out that an apparent blurred interface between the CuO nanorods and the  $\text{Cu}_2\text{O}$  thin film indicates a permeation of the  $\text{Cu}_2\text{O}$  crystalline particles into the CuO nanorods array, revealing the formation of a structure in which the CuO nanorods array is embedded in the  $\text{Cu}_2\text{O}$  thin film.

X-ray diffraction (XRD) was used to verify the formation of CuO and  $\text{Cu}_2\text{O}$  crystalline phases. Figure 3(a) shows a typical XRD pattern of the CuO nanorods/ $\text{Cu}_2\text{O}$  thin film heterostructure fabricated on the ITO-coated glass substrate. The diffraction peaks simply corresponding to either monoclinic tenorite CuO phase (JCPDS card no, 80-1268) or cubic  $\text{Cu}_2\text{O}$  phase (JCPDS card no, 78-2076) were detected. No diffraction peaks of second phases were found in the XRD pattern.

The formation of the CuO/ $\text{Cu}_2\text{O}$  oxide heterostructure can be further confirmed by the Raman spectroscopy analysis. Figure 3(b) shows the Raman spectra of pristine  $\text{Cu}_2\text{O}$  thin film and CuO/ $\text{Cu}_2\text{O}$  heterostructure at room temperature. In these observed spectra, three distinct lines at 150, 220, and  $625\text{ cm}^{-1}$  correspond to the characteristic phonon frequencies of  $\text{Cu}_2\text{O}$  crystalline phase while two distinct lines at 300 and  $350\text{ cm}^{-1}$  correspond to those of CuO crystalline phase [16-18]. For the observed Raman lines from  $\text{Cu}_2\text{O}$  phase, the most intensive line at  $220\text{ cm}^{-1}$  is

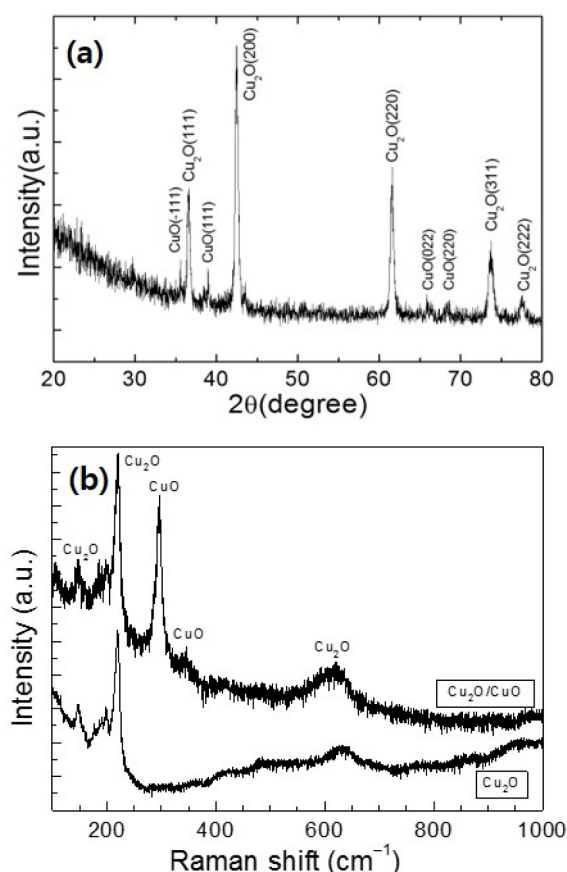


Fig. 3. (a) X-ray diffraction pattern of CuO nanorods/ $\text{Cu}_2\text{O}$  thin film heterostructure, and (b) Raman spectra of  $\text{Cu}_2\text{O}$  thin film and CuO nanorods/ $\text{Cu}_2\text{O}$  thin film heterostructure.

assigned to the  $12\Gamma_{12}^-$  phonon mode, and then the 150 and  $625\text{ cm}^{-1}$  lines are due to the  $\Gamma_{15}^{-(1)}$  and  $\Gamma_{12}^- + \Gamma_{15}^+$  phonon modes, respectively [16,17]. On the other hand, for the Raman lines from CuO phase, the most intensive line at  $300\text{ cm}^{-1}$  is due to Ag phonon mode and the  $350\text{ cm}^{-1}$  line is assigned to Bg phonon mode [18]. Here, it is worth noting that no characteristic Raman lines from CuO phase can be found in the pristine  $\text{Cu}_2\text{O}$  thin film, thus demonstrating the layer electrodeposited upon CuO nanorods array was that of pure  $\text{Cu}_2\text{O}$  phase without Cu or CuO phases.

The PEC water oxidation performance of the fabricated  $\text{Cu}_2\text{O}$  thin film photoelectrode embedded with CuO nanorods was characterized through the current density–potential (J–V) response curves measured using the linear sweep voltammetry in the dark and under light illumination. Figure 4(a) shows the (J–V) response curves of the CuO

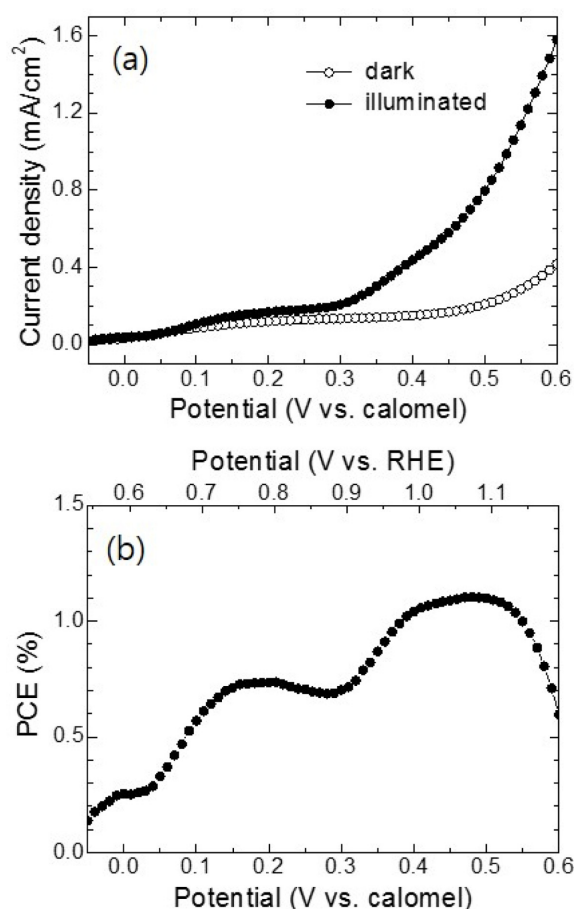


Fig. 4. (a) Current density without and with visible-light illumination, and (b) photocurrent conversion efficiency (PCE) of the CuO nanorods/Cu<sub>2</sub>O thin film heterostructure photoelectrode under an applied potential.

nanorods/Cu<sub>2</sub>O thin film heterostructure photoelectrode in 0.1 M Na<sub>2</sub>SO<sub>4</sub> solution at pH 5.8 in the dark and under light illumination. In the dark, the photoelectrode is clearly seen to exhibit little photocurrent density. However, under light illumination, the CuO/Cu<sub>2</sub>O photoelectrode exhibits a significantly enhanced oxidative photocurrent density, which is mainly ascribed to the PEC water oxidation. For instance, the observed difference in the photocurrent density at the applied potential of 0.5 V vs. calomel is estimated to be 0.59 mA/cm<sup>2</sup>.

The photocurrent conversion efficiency (PCE) for photoelectrochemical water oxidation of the photoanodes were estimated from our observed J–V characteristic data using the following equation [19]

$$\text{PCE}(\%) = \left[ \frac{J \times (1.23 - V_{\text{app}})}{P_i} \right] \times 100$$

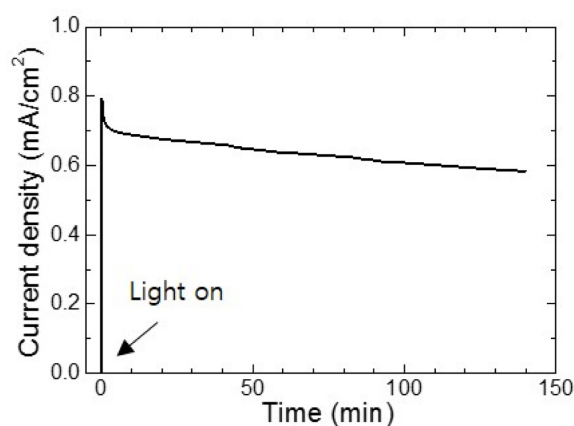


Fig. 5. Photocurrent density-time (J-T) curve measured at 0.5 V versus counter electrode for the the CuO nanorods/Cu<sub>2</sub>O thin film heterostructure photoelectrode under visible-light illumination.

Here,  $V_{\text{app}}$  [V] is the applied external potential vs. a reversible hydrogen electrode (RHE),  $J$  [mA/cm<sup>2</sup>] is the externally measured current density at  $V_{\text{app}}$ ,  $P_i$  [mW/cm<sup>2</sup>] is the power density of the incident light, and 1.23 [V] is the potential for water splitting reaction. The potentials were measured vs. calomel [Hg/Hg<sub>2</sub>Cl<sub>2</sub>/KCl (3.5M)] reference electrode and converted to the RHE scale using the Nernst function

$$E_{\text{RHE}} = E_{\text{Calomel}} + E_{\text{Calomel}}^0 + 0.059\text{pH}$$

Here,  $E_{\text{RHE}}$  is the converted potential vs. RHE,  $E_{\text{Calomel}}$  is the external potential measured against the calomel reference electrode,  $E_{\text{Calomel}}^0$  is the standard electrode potential of the calomel reference electrode (0.250 V vs. RHE at 25°C), and pH is the acidity of the solution. Figure 4(b) presents the change in PCE for the CuO nanorods/Cu<sub>2</sub>O thin film heterostructure photoelectrode under an applied potential vs. calomel and RHE. The CuO/Cu<sub>2</sub>O photoelectrode exhibits the optimal photocurrent conversion efficiency of 1.10% at a potential of 1.06 V vs. RHE (that is, ~0.5 V vs. calomel), which is several times higher than that of Cu<sub>2</sub>O of 0.12% and that of CuO of 0.43% [3]. This electrochemical results convincingly demonstrate that the interlayer of the CuO nanorods array positively influenced the PEC water-splitting



efficiency of the Cu<sub>2</sub>O-based photoelectrodes.

The long-term stability of the CuO/Cu<sub>2</sub>O heterostructure photoelectrode was tested by obtaining the photocurrent density-time (J-t) response curve. An initial photocurrent density of 0.794 mA/cm<sup>2</sup>, obtained by applying 0.5 V between working and counter electrodes, was found to decrease to that of 0.584 mA/cm<sup>2</sup> after 140 min, indicating a considerable decay of ~25% after 140 min and suggest its long-term stability to be addressed.

#### 4. Conclusion

In summary, the CuO nanorods array/CuO thin film bilayered heterostructure was successfully fabricated by a facile method that involved a direct electrodeposition of the Cu<sub>2</sub>O thin film onto the vertically oriented CuO nanorods array to serve as the photoelectrode for the PEC water oxidation. The proposed CuO/Cu<sub>2</sub>O photoelectrode provided an improved performance for the PEC water splitting, indicating good charge separation and transfer efficiency due to the band structure realignment at the interface. The photocurrent density and the optimal photocurrent conversion efficiency obtained on the CuO nanorods/Cu<sub>2</sub>O thin film heterostructure were 0.59 mA/cm<sup>2</sup> and 1.10% at 1.06 V vs. RHE, respectively, which were found to be higher than those of common copper-oxide-based photoelectrodes. This work demonstrate that the Cu<sub>2</sub>O thin film embedded with CuO nanorods could be a promising candidate as a photoanodic material for effective PEC water oxidation. Furthermore, the fabrication strategy of combining electrodeposition and hydrothermal synthesis provides a facile, low-cost, and scalable approach towards to preparing earth-abundant copper-oxide-based photoelectrodes for visible-light-driven hydrogen generation.

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