

Forming Gas Post Metallization Annealing of Recessed AlGaIn/GaN-on-Si MOSHFET

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Abstract—In this study, the effects of forming gas post metallization annealing (PMA) on recessed AlGaIn/GaN-on-Si MOSHFET were investigated. The device employed an ICPCVD SiO₂ film as a gate oxide layer on which a Ni/Au gate was evaporated. The PMA process was carried out at 350°C in forming gas ambient. It was found that the device instability was improved with significant reduction in interface trap density by forming gas PMA.

Index Terms—AlGaIn/GaN heterostructure, forming gas annealing, interface trap density, MOSHFET, post metallization annealing

I. INTRODUCTION

AlGaIn/GaN heterostructures are a promising candidate for use in high-power and high-efficiency switching applications owing to their superior material properties, such as high breakdown field and high electron mobility [1-4]. In particular, AlGaIn/GaN wafers grown on silicon substrates have gained additional attention due to their cost competitiveness with current Si technology. Prototype AlGaIn/GaN FETs and power ICs have already demonstrated very low on-resistance and superior conversion efficiency in comparison with the state-of-the-art Si counterparts [5-7].

Insulated gate configuration has been widely

employed in AlGaIn/GaN power FETs because the insulated gate can suppress the off-state leakage current and adjust the threshold voltage. However, the bulk and trap charges generated during deposition and following process steps have strong influence on device characteristics, which becomes more critical when the device employs recessed-gate configuration with *ex-situ* gate insulator deposition. Since surface and interface states are strongly process-dependent, the gate insulator process must be carefully optimized.

It was reported that the insulator itself and interface conditions between insulator and semiconductor surface could be improved by appropriate post-annealing processes [8-12]. In this study, we investigated the effects of forming gas post metallization annealing (PMA) on recessed AlGaIn/GaN-on-Si MOSHFET with ICPCVD SiO₂ gate oxide.

II. EXPERIMENTS AND DISCUSSIONS

Fig. 1 shows the cross-sectional schematic of recessed AlGaIn/GaN-on-Si MOSHFET. The epitaxial layer structure consisted of a 4 nm undoped GaN capping layer, a 20 nm undoped AlGaIn barrier, a 5 μm undoped GaN buffer on an N-type Si (111) substrate. The recessed gate region formed the metal gate/SiO₂/GaN MOS configuration whereas the parasitic channel region had the AlGaIn/GaN heterostructure to keep a low channel resistance. The thickness of SiO₂ gate oxide was 35 nm. The 1 μm overhang from the gate edge formed a field plate under which a 150 nm SiN_x layer was inserted. The source-to-gate distance, gate length, and gate-to-drain distance were 3, 2, and 15 μm, respectively.

Devices were fabricated using the following process

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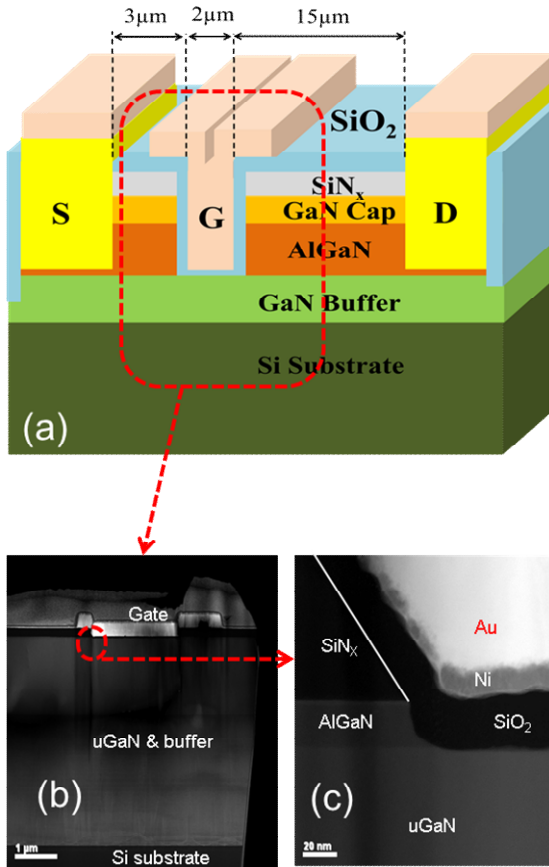


Fig. 1. (a) Cross-sectional schematic of recessed AlGaIn/GaN MOSFET, (b, c) TEM images of the recessed MOS gate region.

steps. After cleaning the wafer, a 150 nm SiN_x film, as a prepassivation layer, was deposited at 250°C using ICPCVD. Mesa isolation and gate recess were carried out using Cl₂/BCl₃ based ICP-RIE. As shown in Fig. 1(c), a sloped gate recess profile was obtained to suppress the localized high electric field. Ohmic contacts were formed by Si/Ti/Al/Mo/Au metallization followed by RTA annealing at 830°C. A 35 nm SiO₂ film was deposited at 250°C using ICPCVD [13]. A Ni/Au layer was evaporated for the gate and pad regions. Lastly, PMA was carried out by RTA at 350°C for 3 min in forming gas ambient (H₂(5%) + N₂(95%)) with the chamber pressure of 5 Torr.

1. Current-Voltage Characteristics

Current-voltage (I-V) transfer measurements were repeated using a dual sweep mode, i.e. positive and negative directions. As shown in Fig. 2, the initial characteristics before PMA process exhibited significant

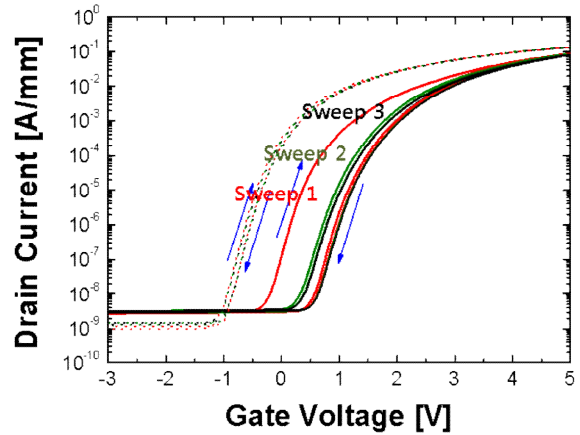


Fig. 2. Hysteresis of I-V transfer characteristics before (solid lines) and after (dot lines) forming gas PMA. The first, second, and third sweeps are shown in red, green, and black, respectively.

hysteresis with memory effects, suggesting negatively charged trapping effects. In contrast, no hysteresis was observed with a slightly enhanced on/off ratio after forming gas PMA. It should be noted that negative shift in threshold voltage was observed after forming gas PMA. In order to take a closer look at threshold voltage shift and uniformity issues, about 30 devices were measured at various locations in the sample. The threshold voltage variations before and after PMA are compared in Fig. 3. The initial threshold voltage was widely distributed from 1 to 4 V before PMA. On the other hand, the variation was significantly reduced to the range between -0.2 and 0.7 V after PMA. Since the threshold voltage is a strong function of oxide and interface charges, the threshold voltage and its uniformity must be associated with the change in oxide/interface charges during forming gas PMA. It is speculated that large variation of threshold voltage before PMA implies non-uniform distribution of oxide bulk charges caused by oxygen vacancies, and the hydrogenated dangling bonds and reduced oxygen vacancies after forming gas PMA are responsible for the improved uniformity and stability. Since the hydrogenated dangling bonds in SiO₂ act as positive charges, they would make negative shift in threshold voltage [14].

In terms of breakdown voltage characteristics, no significant change was observed after the forming gas PMA. Typical off-state breakdown voltages both before and after PMA were at least 1000 V.

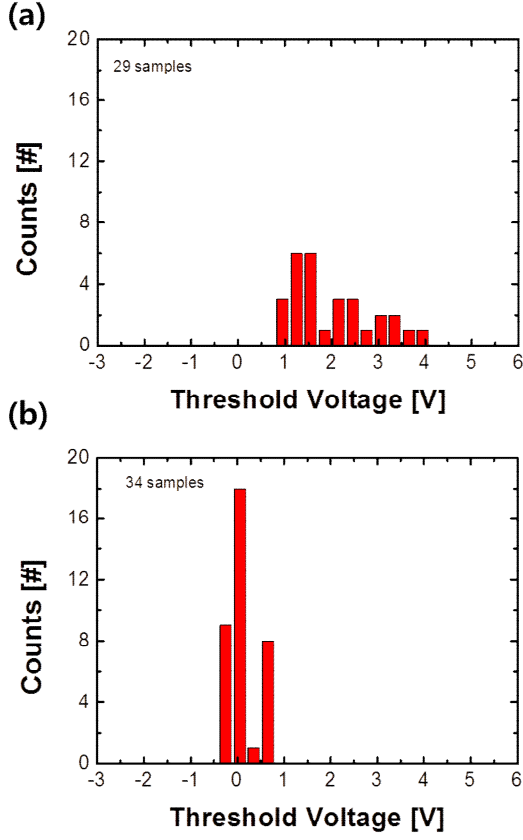


Fig. 3. Threshold voltage variation (a) before, (b) after forming gas PMA.

2. Capacitance-Voltage Characteristics and Interface Trap Density Extraction

In order to investigate the interface quality, the capacitance-voltage (C-V) characteristics were measured before and after PMA. C-V measurements were carried out using a circular recessed C-V pattern with a diameter of 50 μm . As compared in Fig. 4(a), the C-V hysteresis at 1 MHz was significantly reduced from 800 to 150 mV after forming gas PMA.

In order to quantitatively analyze the interface conditions, two different extraction methods, i.e. Terman [15] and Conductance [16] methods, were employed to estimate the interface trap density before and after PMA.

In Terman method, the interface trap density (D_{it}) is extracted by the following equation [17, 18]

$$D_{it} = \frac{C_{ox}}{q} \left[\frac{\left(\frac{\partial C}{\partial V} \right)_{ideal}}{\left(\frac{\partial C}{\partial V} \right)_{exp}} - 1 \right] \quad (1)$$

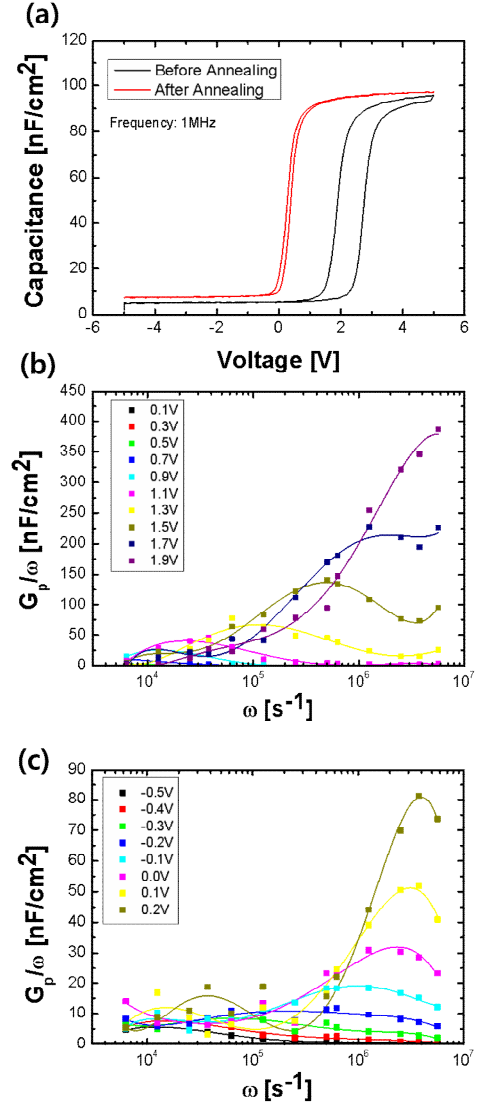


Fig. 4. (a) C-V hysteresis characteristics at 1 MHz and G_p/ω versus ω , (b) before, (c) after forming gas PMA.

where C_{ox} is the oxide capacitance and q is the electronic charge. The ideal capacitance characteristics were obtained by Schroder's method [19].

Unlike Terman method, Conductance method utilizes frequency dependent C-V characteristics to estimate D_{it} . The C-V characteristics were measured from 1 kHz to 1 MHz from which normalized conductance values were calculated as plotted in Figs. 4(b) and (c). The relationship between the normalized conductance (G_p/ω) and D_{it} is given by [19]

$$\frac{G_p}{\omega} = \frac{\omega G_m C_{ox}^2}{G_m^2 + \omega^2 (C_{ox} - C_m)^2} \quad (2a)$$

$$\frac{G_p}{\omega} = \frac{qD_{it}}{2\omega\tau_{it}} \ln \left[1 + (\omega\tau_{it})^2 \right] \quad (2b)$$

where G_p is the parallel conductance, ω is $2\pi f$ (f = measurement frequency), G_m is the measured conductance, C_m is the measured capacitance, C_{ox} is the oxide capacitance, and τ_{it} is the trap time constant. An appropriate expression giving D_{it} in terms of the measured maximum conductance is $D_{it} \approx 2.5(G_p/q\omega)$ when $\omega \approx 2/\tau_{it}$ [19, 20]. The relationship between τ_{it} and the trap state energy (E_c-E_t) is given by [21]

$$\tau_{it} = \left[(\sigma_{it}N_c\nu_T)^{-1} \exp\left(\frac{E_c-E_t}{kT}\right) \right] \quad (3)$$

where σ_{it} is the capture cross section of trap states, N_c is the density of states in the conduction band, and ν_T is thermal velocity. The values used for σ_{it} , N_c , and ν_T in this work are $3.4 \times 10^{-15} \text{ cm}^2$, $4.3 \times 10^{14} \times T^{3/2} \text{ cm}^{-3}$, and $2.6 \times 10^7 \text{ cm/s}$, respectively [21, 22].

The D_{it} values extracted using Terman and Conductance methods are plotted in Fig. 5. Although two different methods did not give the same D_{it} values, both methods clearly exhibited noticeable reduction in D_{it} values after forming gas PMA. In comparison with other reports for GaN MIS configuration [23-25], very low D_{it} values were achieved in this work. It is suggested that the reduced interface trap states after forming gas PMA are responsible for the reduced hysteresis phenomenon.

III. CONCLUSION

We investigated the effects of forming gas PMA on recessed AlGaIn/GaN-on-Si MOSHFETs that employed a SiO₂ gate oxide. It was found that the forming gas PMA not only eliminated the hysteresis phenomenon but also improved the threshold voltage uniformity. It is suggested that the hydrogenated dangling bonds and reduced oxygen vacancy in conjunction with the improved interface conditions are responsible for the improved device characteristics.

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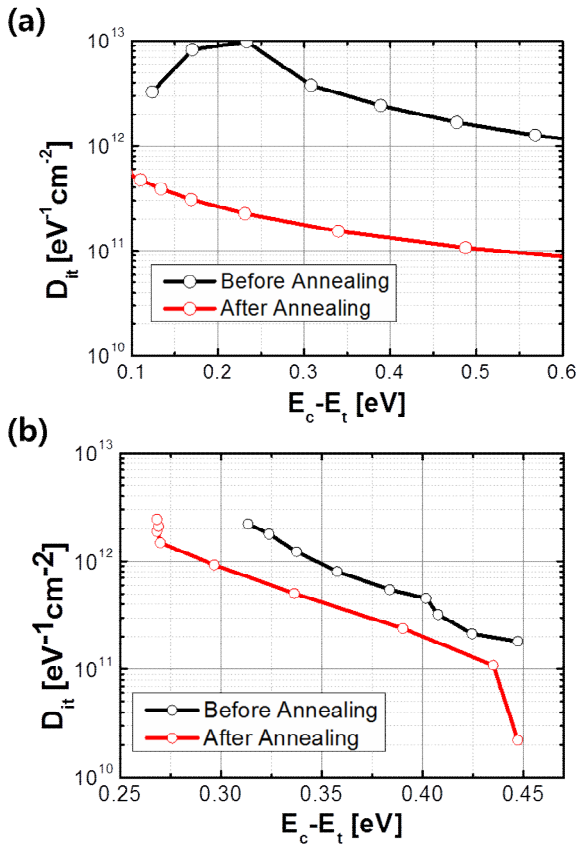


Fig. 5. D_{it} characteristics before and after forming gas PMA extracted by (a) Terman, (b) Conductance methods.

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