



# Ferroelectric Phase Transition of Lead Free $(1-x)(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3-x\text{LiNbO}_3$ Ceramics

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Lead-free  $(1-x)(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3-x\text{LiNbO}_3$ , i.e., NKN-LN $x$  ( $x=0.0, 0.1, 0.2, 0.3, 0.4$  mol) was prepared using the conventional solid state reaction method. The effects of LN mixing on the ferroelectric properties of NKN-LN $x$  ceramics were studied using a dielectric constant and  $P$ - $E$  (Polarization-electric field) measurements. Ferroelectricity was observed in the composition for  $x$  approximately varying between 0.0 and 0.4. Minimum remanent polarization  $2P_r=5$  C/cm<sup>2</sup> was achieved in the composition for  $x=0.2$ . The ferroelectric phase transition temperature  $T_c$  increased with increasing LN content. The ferroelectric phase transition of NKN-LN $x$  ( $x \geq 0.1$ ) is a second-order phase transition, and that of NKN-LN $x$  ( $x \leq 0.2$ ) is a first-order phase transition. These results indicate that the ferroelectric phase transition temperature of NKN-LN $x$  change from that of second-order to weak first-order phase transition according to the LN content.

**Keywords:** Lead-free, Oxides, Impedance spectroscopy, Dielectric, Phase transition

## 1. INTRODUCTION

In the field of piezoelectric ceramics, sodium potassium niobate ceramics with lead-free piezoelectric material have been investigated as alternative material for PZT-based ceramics [1-13]. Lead-free ferroelectric materials with perovskite structure have a general formula of  $\text{ABO}_3$ . In this structure, cations based on their valence states and coordination numbers occupy the A- or B-sites.  $\text{Na}_{1-y}\text{K}_y\text{NbO}_3$ , NKN is a material with perovskite structure, and it exhibits high piezoelectric properties because its structure permits spontaneous polarization to rotate along three orientations. Sodium potassium niobate (NKN) is a solid solution of potassium niobate (KN) a ferroelectric and sodium niobate (NN), with an Na/K ratio of  $\sim 50/50$ . The piezoelectric applications of

$\text{Na}_{0.5}\text{K}_{0.5}\text{NbO}_3$  (NKN), ceramics produced by hot-pressing, are better than those produced by sintering in air atmosphere. Hot-pressed NKN ceramics have been reported to have a high phase transition temperature ( $T_c \sim 420^\circ\text{C}$ ), good piezoelectric properties ( $d_{33} \sim 160$  pC/N), and a high planar coupling coefficient ( $\kappa_p \sim 45\%$ ) [1-4].

However, NKN ceramics are difficult to obtain using the conventional sintering method because their phase stability is limited to  $1,140^\circ\text{C}$  and they are exposed to moisture. Therefore, attempts have been made to improve the sinterability and piezoelectric properties of KNN through the addition and/or substitution of several cationic elements in the A- or B-sites [10-13]

It is known that  $(1-x)(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3-x\text{LiNbO}_3$ , NKN-LN $x$  ceramics are good, lead-free piezoelectric and ferroelectric ceramics. A morphotropic phase boundary between the orthorhombic phase and the tetragonal phase of NKN-LN $x$  was present when  $x$  was approximately 0.05  $\sim$  0.07 mol of LN [8]. Guo *et al.*, observed that the Curie temperatures ( $T_c$ ) of NKN-LN $x$  ceramics were in the range of  $452 \sim 510^\circ\text{C}$ , according to their LN content, which is at least  $100^\circ\text{C}$  higher than that of  $\text{Pb}(\text{Zr}, \text{Ti})\text{O}_3$ . For  $(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3$ ,

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$T_c$  values were observed at 420 °C and 200 °C, which correspond to the cubic-orthorhombic and orthorhombic-tetragonal phase transitions, respectively. Two phase transitions were present at  $x = 0.04, 0.06$  mol, similar to the case for NKN, except that the phase transition temperatures were shifted [9]. Many research efforts thus far have been based on the conditions for which a small amount of LN was added to the NKN composition.

In this study,  $(1-x)(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3-x\text{LiNbO}_3$ , i.e., NKN-LN $x$  ( $x = 0.0, 0.1, 0.2, 0.3, 0.4$  mol), was synthesized using the conventional solid state method. The purpose of this study is to investigate the phase transition and electrical properties of  $(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3$  in terms of its  $\text{LiNbO}_3$  content.

## 2. EXPERIMENTS

Lead-free  $(1-x)(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3-x\text{LiNbO}_3$ , i.e., NKN-LN $x$  ( $x = 0.0, 0.1, 0.2, 0.3, 0.4$  mol), was prepared by mixing the oxides,  $\text{K}_2\text{CO}_3$  (99% purity),  $\text{Na}_2\text{CO}_3$  (99% purity),  $\text{LiNbO}_3$  (99% purity) and  $\text{Nb}_2\text{O}_5$  (99% purity) in a molar ratio used in the conventional solid state reaction method. Before being weighed, the  $\text{K}_2\text{CO}_3$  and  $\text{Na}_2\text{CO}_3$  powders were first dried in an oven at 200 °C for 10 h to minimize the effect of moisture. These powders were then milled with  $\text{ZrO}_2$  balls for 20 h using ethyl alcohol as a medium and dried. The dried powders were calcined at 850 °C for 2 h. After calcination, the powders were ball-milled again for 20 h and, dried, after which PVA (4 wt%) was added as a binder. They were then pressed into disks with diameter of under 13 mm. After burning off the PVA, the pellets were sintered at 1,070 °C for 2 h. The crystal structures were determined by X-ray powder diffraction analysis using  $\text{CuK}\alpha$  radiation (Philips X'Pert - MPD system). The remnant polarization  $P_r$  and coercive field  $E_c$  were determined from the P-E (Polarization - Electric field) hysteresis loops, as measured by a Radiant Precision Workstation. To examine their dielectric properties, the ceramics were polished and painted with silver paste on both surfaces, and fired at 800 °C for 30 min. The real and imaginary dielectric constants were measured using an SI1260 impedance analyzer at temperature ranging from room temperature to ~ 600 °C with heating and cooling rates of 0.2 °C/min in the frequency range of 1 Hz to 1 MHz.

## 3. RESULTS AND DISCUSSION

Figure 1 shows the XRD patterns of the  $(1-x)(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3-x\text{LiNbO}_3$ , i.e., NKN-LN $x$  ( $x = 0.0, 0.1, 0.2, 0.3, 0.4$  mol) ceramics. Studies have reported that a phase of  $\text{K}_3\text{Li}_2\text{Nb}_5\text{O}_{15}$  (KLN) with a tetragonal tungsten bronze structure starts to appear at  $x \geq 0.08$  [9]. In this study, it appeared at  $x \leq 0.2$  but for  $x \geq 0.3$ , the KLN phase and  $\text{LiNbO}_3$  phase coexisted. This implies that the structures of the NKN-LN $x$  ceramics were transformed, again increasing their  $\text{LiNbO}_3$  content.

P-E hysteresis loops of  $(1-x)(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3-x\text{LiNbO}_3$ , i.e., NKN-LN $x$  ( $x = 0.0, 0.1, 0.2, 0.3, 0.4$  mol) ceramics measured at room temperature under a driven electric field are plotted in Figs. 2(a)-(f). Generally, the presence of P-E hysteresis loops is considered to be evidence that a material is ferroelectric.

The capacitor is characterized by P-E hysteresis curves. However, the shapes of the P-E loops changed slightly with increasing LN contents. As shown in Fig. 2(f), the value of  $2P_r$  decreases with an increasing LN content below a certain critical level.  $2P_r$  has a minimum value of 5 C/cm<sup>2</sup> near  $x = 0.2$ , and it first increases and then decreases after reaching this value. The coercive field  $2E_c$  increases for an increase in the amount of LN in the range between  $x = 0.0$  and  $x = 0.1$  mol., and a further increase in the amount of LN above  $x = 0.2$  mol causes an increase in  $2E_c$ .

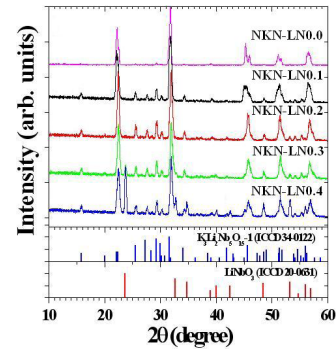


Fig. 1. X-ray diffraction patterns of the  $(1-x)(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3-x\text{LiNbO}_3$ , NKN-LN $x$  ceramics.

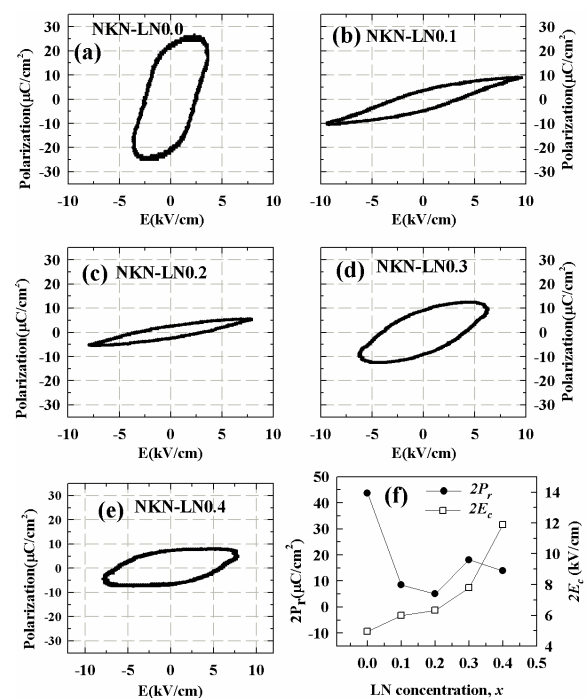


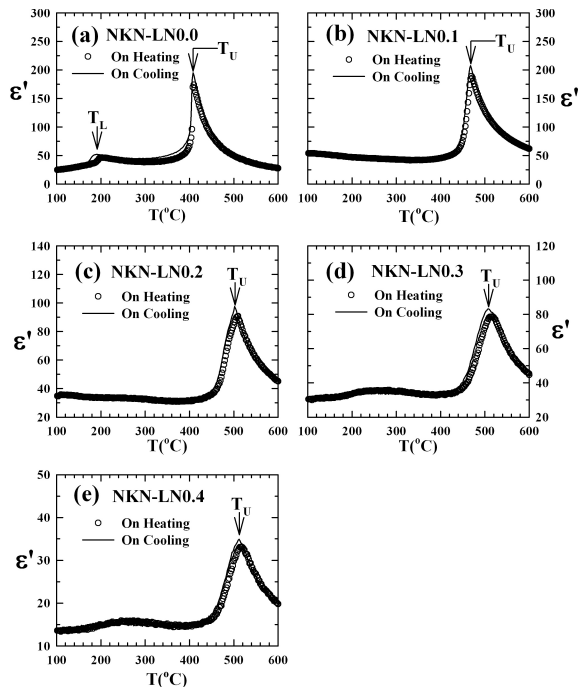
Fig. 2. Ferroelectric hysteresis loops of the  $(1-x)(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3-x\text{LiNbO}_3$ , NKN-LN $x$  ceramics for (a)  $x = 0.0$ , (b)  $x = 0.1$ , (c)  $x = 0.2$ , (d)  $x = 0.3$ , and (e)  $x = 0.4$  mol, (f) remnant polarization and coercive field of NKN-LN $x$  ceramics as a function of the LN contents  $x$ .

The tendency of varying  $2P_r$  is similar to that of  $2E_c$  when the range of  $x$  is approximately above  $x = 0.2$  mol.

Du *et al.* [8] reported the dielectric properties of NKN-LN $x$  ceramics for the case that the amount of LN is below  $x = 0.2$  mol; when the amount LN is  $x = 0.06$  mol,  $E_c$  achieves its minimum value of 13.4 kV/cm and  $P_r$  reaches its minimum value of 20 C/cm<sup>2</sup>. They proposed that NKN-LN0.06 ceramics are a promising candidate for lead-free high-temperature piezoelectric ceramics.

Figures 3(a) and (e) show the real ( $\epsilon'$ ) dielectric constant at 1 MHz as a function of temperature for  $(1-x)(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3-x\text{LiNbO}_3$ , i.e., NKN-LN $x$  ( $x = 0.0, 0.1, 0.2, 0.3, 0.4$  mol) ceramics. In the case of NKN-LN0.0 ceramics, the values of  $\epsilon'$  increase with decreasing temperature. At  $T_c$  (the temperature at which  $\epsilon'$  is maximized) = 409 °C,  $\epsilon'$  begins to decrease, forming a large  $\lambda$ -type peak in the dielectric constant *vs.* temperature curve upon heating and cooling.

As the temperature decreases, if we assume that the phase



**Fig. 3.** The temperature dependence of the real dielectric constant  $\epsilon'$  in  $(1-x)(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3-x\text{LiNbO}_3$ , NKN-LN $x$  ceramics at 1 MHz on heating (symbol) and cooling (solid line), (a)  $x=0.0$ , (b)  $x=0.1$ , (c)  $x=0.2$ , (d)  $x=0.3$ , and (e)  $x=0.4$  mol.

transition temperature is the mid-point of the steepest curve of  $\epsilon'$ , then the lower transition occurs at  $T_{\text{OTC}}$  (low temperature phase transition point) = 176 °C upon cooling and at  $T_{\text{OTH}}=195$  °C upon heating with a thermal hysteresis of 19 °C. This result is similar to that reported by Guo *et al.* [9].

In the case of NKN-LN0.1, a low temperature anomaly was not observed at  $T_{\text{OT}}$  upon heating or cooling.

At high temperatures, the complex dielectric response of NKN-LN0.1 was found to be similar to that of NKN-LN0.0. The sharp peaks around  $T_C$  for the NKN-LN0.0 and NKN-LN0.1 samples show a second-order phase transition without thermal hysteresis.

In the case of NKN-LN $x$  ( $x \geq 0.2$ ), the ferroelectric phase transition temperature  $T_C$  shifted to a higher value with an increase in the LN content, whereas the dielectric peak broadened. The temperature anomaly of the real dielectric constant appeared at  $T_C$  in all the samples upon heating and cooling with a small thermal hysteresis, which corresponded to a weak first-order phase transition. A low-temperature dielectric anomaly was not observed upon heating and cooling. In NKN-LN $x$  samples with  $0 \leq x \leq 0.07$ , Guo *et al.* [9] reported that the phase transition of NKN-LN0.0 was observed at 420 °C and 200 °C, which corresponds to the cubic-orthorhombic (at  $T_C$ ) and orthorhombic-tetragonal (at  $T_{\text{OT}}$ ) phase transitions. Also,  $\text{LiNbO}_3$  has lithium niobate structure, which can be described as a heavily distorted perovskite or an ordered phase derived from the corundum structure with space group  $R_{3c}$  ( $C_{3v}^6$ ). So, it is evident that two effects on the structure of NKN ceramics have been observed in NKN-LiNbO $_3$  ceramics. At lower  $\text{LiNbO}_3$  concentrations, Li mainly replaces Na and K in the A sites of ABO $_3$  perovskite structure (i.e. form a solid solution), leading to a linear shift of the Curie point ( $T_C$ ) to higher temperature [9]. However, the structure of solid solution transforms from orthorhombic to tetragonal symmetry due to the large distortion caused by  $\text{Li}^+$  [9].

The phase transition temperatures also shifted increasing the

**Table 1.** Phase transition temperature ( $T_{\text{OT}}$ ,  $T_C$ ) of NKN-LN $x$  ceramics on heating and cooling. unit: °C

Samples	On Heating		On Cooling		$\Delta T_{\text{OT}}$	$\Delta T_C$
	$T_{\text{OTH}}$	$T_{\text{OTC}}$	$T_{\text{CH}}$	$T_{\text{CC}}$		
NKN-LN0.0	195	176	409	409	19	0
NKN-LN0.1			469	469		0
NKN-LN0.2			509	502		7
NKN-LN0.3			517	508		9
NKN-LN0.4			521	514		7

LN content.  $T_C$  shifted to a higher value, and  $T_{\text{OT}}$  to a lower value [11]. Thus, we expect that a low-temperature phase transition of this sample should appear at room temperature because these phase transition temperatures decrease with an increase in LN contents.

The values of  $T_{\text{OT}}$ ,  $T_C$ , and  $\Delta T$  obtained for all the samples are presented in Table 1. Here,  $\Delta T$  indicates the degree of the first- and second-order phase transition of NKN-LN $x$ . These results indicate that the phase transition of NKN-LN $x$  ceramics occurs when  $T_C$  changes from a second-order to weak first-order phase transition with increasing LN contents. Our results also show the possibility that the concentration of  $x = 0.2$  may be the critical concentration for a first- to second-order-ferroelectric phase transition.

## 4. CONCLUSIONS

In conclusion,  $(1-x)(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3-x\text{LiNbO}_3$ , i.e., NKN-LN $x$  ( $x=0.0, 0.1, 0.2, 0.3, 0.4$ mol) ceramics, were synthesized using the solid state reaction method. The effects of LN mixing on the ferroelectric properties of these two ceramics were studied through dielectric and P-E measurements. The value of  $P_r$  increased with increasing Nb content.  $(1-x)(\text{Na}_{0.5}\text{K}_{0.5})\text{NbO}_3-x\text{LiNbO}_3$  ceramics exhibited a minimum remanent polarization of  $2P_r=5$   $\mu\text{C}/\text{cm}^2$  at an LN content of  $x \sim 0.2$ . These results indicate that LN doping can change the ferroelectric properties of NKN-LN $x$  ceramics. The phase transition temperature,  $T_C$ , increased with increasing LN contents. The ferroelectric phase transition of NKN-LN $x$  ( $x \leq 0.1$ ), is a second-order transition without thermal hysteresis, and NKN-LN $x$  ( $x \geq 0.2$ ) is a weak first-order transition with small thermal hysteresis. Thus, our results demonstrate the possibility that the concentration of  $x \sim 0.2$  may be the critical concentration for a first-to-second-order-ferroelectric phase transition.

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