Preparation and Electrochemical Performance

Preparation and Electrochemical Performance of CNT Electrode with Deposited Titanium Dioxide for Electrochemical Capacitor

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To reduce polarization of electrochemical capacitor based on carbon nanotube, titanium oxide nanoparticles were deposited by ultrasound. The pore distribution of TiO₂/CNT nanoparticle exhibited surface area of 341 m²g⁻¹ when TiO₂ content was 4 wt %, which was better than that of pristine CNT with surface area of 188 m²g⁻¹. The analyses indicated that titanium oxide (particle diameter < 20 nm) was deposited on the CNT surface. The electrochemical performance was evaluated by using cyclic voltammetry (CV), impedance measurement, and constant-current charge/discharge cycling techniques. The TiO₂/CNT composite electrode showed relatively better electrochemical behaviors than CNT electrode by increasing the specific capacitance from 22 Fg⁻¹ to 37 Fg⁻¹ in 1 M H₂SO₄ solution. A symmetric cell assembled with the composite electrodes showed the specific capacitance value of 11 Fg⁻¹ at a current loading of 0.5 mAcm⁻² during initial cycling.

Key Words: Electrochemical capacitor, Titanium dioxide, CNT, Polarization

Introduction

Electrochemical capacitor is an electrochemical energy storage device utilizing the electric charges accumulated at the interface between the electrode and the electrolyte to form the electric double layer.¹⁻⁵ It has been increasingly getting attention not only for the established applications as backup power to electronic equipments and mobile devices,^{6,7} but also for high power applications in pulsed lasers and electric vehicles.^{8,9}

Advancement in energy-storage technologies can benefit from the shift from conventional to nanostructured electrodes. For electrochemical capacitors, focus has been on nanostructured carbons, nanotubes, and nanotemplates.^{10,11} Among these materials, carbon nanotubes (CNTs) have been considered ideal for electrochemical capacitors, owing to the advantages such as accessible surface area, excellent electronic conductivity, and good stability.¹²⁻¹⁷

On the other hand, CNT grows the polarization when CNT electrode reacts with electrolyte through the electrode/electrolyte interface. Generally, polarization is the change of potential of an electrode from its equilibrium potential upon the application of a current. The phenomenon of transportation of electrical charge from one part of the electrochemical cell to another, occurring mainly as electromigration of ions, but it can also occur by diffusion of ions.^{5,17} The specific capacitance of the electrochemical capacitor based on CNTs has been reported to present lower capacitance of about $20 \sim 30 \text{ Fg}^{-1}$. The lower capacitance is partially attributed to poor wettability of electrode material, which leads to a lower usable specific surface area for charge storage.¹⁸ As another reason, polarization may reduce charge accumulation of the ions on the double-layer. For these reasons, reducing the polarization of activated carbon by intro-

ducing TiO_2 nanoparticles has been considered to be an effective approach to improve the capacitance of the double-layer capacitors.⁵ In this study, we considered that wettability and polarization of the nanocomposite electrode depend on the content of TiO_2 deposited on CNT.

Experimental

The TiO₂/CNT nanocomposite material was prepared by solgel process with ultrasound in 2-propanol. The starting material, titanium isopropoxide (Aldrich Co., Ltd.) was commercially available with purity of 99%. The average outer diameter of CNT (MWNT type, Iljinnanotech. Co., Ltd.) was 15 ~ 20 nm and the length was $10 \sim 20 \,\mu\text{m}$. The CNT was chemically oxidized in 1M H₂SO₄ for 48 h at 80 °C and dried at 120 °C overnight. Titanium isopropoxide (0.2, 0.4, 0.6 and 0.8 M) and 1 g CNTs were mixed in 2-propanol (200 mL) with ultrasound, respectively. Then they were mixed with stirring under ultrasound for 30 min. The mixture was separated from the solvent by drying at 90 °C into vacuum oven overnight. The resultant was calcinated at 450 °C for 2 h in air at a heating rate of 5 °C/min to form TiO₂/CNT nanocomposite. The samples were characterized by X-ray diffraction (XRD, SCINTAG DMS2000) and transmission electron microscope (TEM, model Joel-1020 Carl Zeiss TEM109) was used to monitor the particle size at each synthesis step. X-ray Photoelectron Spectroscopy (XPS, ESCALAB 210) was used to obtain the structure of nanocomposite.

The working electrode was prepared from a mixture paste of TiO₂/CNT nanocomposite powder, ketijen black(KB) as an electronic conductor additive, and polyvinylidene fluoride (PVdF) binder mixed with the weight ratio of 85:10:5 in *N*-Methyl-2-Pyrrolidone (NMP). The electrode was dried under vacuum at 120 °C for 24 h before electrochemical evaluation. Electrochemical measurements were conducted using conventional three-electrode configuration and symmetrical full cell. The TiO₂/CNT composite on carbon paper electrode was used as the working electrode. A Pt wire and KCl-saturated Ag/AgCl were used as the counter and reference electrodes, respectively. The 1M H₂SO₄ aqueous solution was used as the electrolyte. Cyclic voltammetry (CV) was measured on range of 0 ~ 800 mV. Scan rate was 50mVs⁻¹ and it was shown after 3 th cycle. And cycling performance of the two electrodes in cell was measured at 0.5 mAcm⁻² between 0 and 1000 mV. All the measurements were taken at room temperature and under nitrogen atmosphere.

Results and Discussion

In order to show actual size and formation of deposited TiO₂ particle, Fig. 1 (b) ~ (e) present TEM images of TiO₂/CNT composite materials with different TiO₂ weight ratio. Some of the shapes of these samples were classified as a part of a straight form with a thick outer diameter and another rest as a curved style with a narrow outer diameter. The TiO₂/CNT composite materials were different from the result in Fig. 1. From above results, our attempt to prepare the TiO₂/CNT composite materials was successful. And the more the TiO₂ was inserted, the more the deposited TiO₂ grains clumped together. The particles are spherical with the size of about 10 ~ 20 nm in diameter.

Fig. 2 shows the X-ray diffraction (XRD) patterns of the CNT and the CNT on which TiO2 was deposited with contents of 4, 6, 17, and 32 wt %. After the stoichiometric precursor of $TiO_2/$ CNT composite has been calcinated at 450 °C for 2 h in air, the anatase phase of TiO₂ has been appeared. The peaks located at 25.66° and 42.50° could be identified to the reflection from the (002) and (100) planes of CNT and the additional peaks observed at 25.07°, 37.66°, 47.74°, 53.8°, 54.29° and 62.44° are corresponding to (101), (004), (200), (105), (211) and (204) planes of anatase TiO₂ (JCPDS no. 21-1272).¹⁹ All of broaden and weaken peaks in Fig. 2 imply that TiO₂/CNT composite materials consisted of TiO₂ and CNT nanoparticles belong to low-dimensional states. The CNT (002) reflection overlaps the anatase $TiO_2(101)$ reflection. As is seen in Fig. 2, the TiO_2/CNT composite shows CNT peak, and also anatase peaks of the composites grow as the TiO₂ content increases. And no peak of other phases was observed, which indicates that the composite materials are well crystallized.

The X-ray photoelectron spectroscopy (XPS) which was performed to investigate the chemical bonding state of Ti in the TiO₂/CNT composite material prepared at 450 °C for 2 h is shown in Fig. 3. The Ti 2p signal of the TiO₂/CNT composites shows a Ti 2p_{1/2} peak centered at 464.8 eV and a Ti 2p_{3/2} peak at 459 eV. The observed binding energies (B.E.) of Ti 2p_{1/2} and Ti 2p_{3/2} peaks for TiO₂/CNT composites are near to those observed for TiO₂ (458.5 and 464.2 eV, respectively). The atomic % of TiO₂ nanoparticle (C_x) was calculated peak area from the equation (1):

$$C_{\rm x} \text{ atomic } \% = \left[(I_i / F_i) / \Sigma (I / F) \right] \times 100 \tag{1}$$

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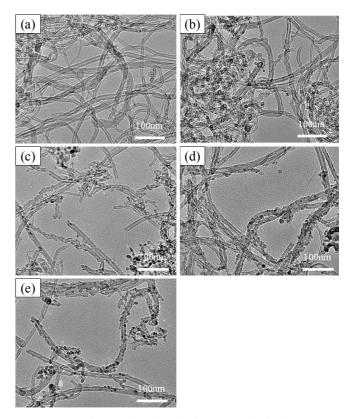


Figure 1. TEM images of (a) the pristine CNT and TiO_2/CNT nanocomposites with deposited TiO_2 contents of (b) 4 wt %, (c) 6 wt %, (d) 17 wt % and (e) 32 wt %.

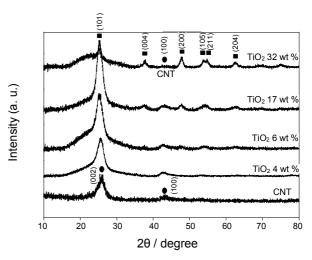


Figure 2. X-ray diffraction patterns (Cu K α) of the pristine CNT and TiO₂/CNT nanocomposites with various TiO₂ contents.

where *I* is the number of photoelectron per second in a specific spectra peak, *F* is the atomic sensitivity factor, *i* is the TiO₂ element. According to the XPS results, TiO₂/CNT composites contained TiO₂ particle of correctly 4 wt %, 6 wt %, 17 wt % and 32 wt % on CNT, respectively.

Fig. 4 shows the nitrogen adsorption isotherms of the (a) CNT and the CNT on which TiO_2 was deposited with contents of (b) 4, (c) 6, (d) 17, and (e) 32 wt %. An increase in the TiO_2 con-

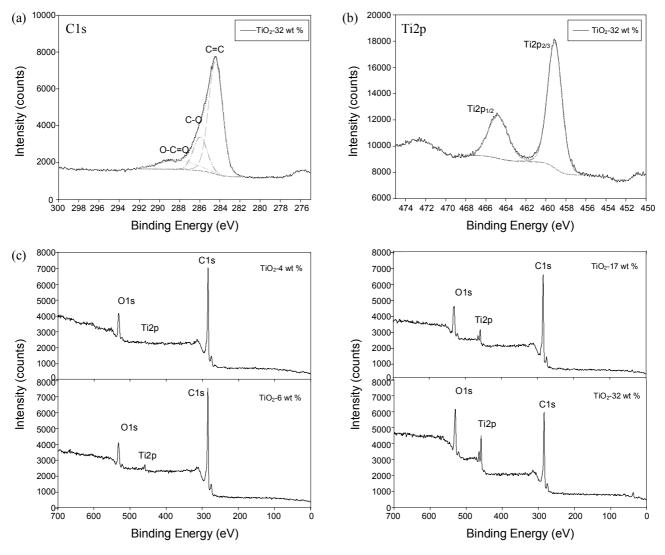


Figure 3. The X-ray photoelectron spectroscopic scans of the pristine CNT and TiO_2/CNT nanocomposites with various TiO_2 contents. (a) C1s (32 wt %), (b) Ti2p (32 wt %), and (c) wide scans of TiO_2/CNT nanocomposites.

tents a great enhancement in the nitrogen adsorption capacity. Fig. 5 shows the BET surface area of the pristine CNT and TiO₂/CNT nanocomposites with various deposited TiO₂ contents. After deposition of TiO₂, surface area raises the porosity of the TiO₂/CNT composite to improve BET specific surface area. When comparing those TiO₂/CNT composites, 4 wt % TiO₂/CNT composites and 6 wt % TiO₂/CNT composites have higher specific surface area. The pore distribution of TiO₂/CNT nanocomposites is exhibited in Table 1. As seen in this Table, the 4 wt % and 6 wt % TiO₂/CNT composites showed most excellent values of specific surface area (341 m²g⁻¹) and micro pore area (17 m²g⁻¹) compared with the pristine CNT, respectively. But higher contents of deposited TiO₂ decreased both specific surface area and micro pore area.

The effect of TiO₂ on the capacitance property of the TiO₂/ CNT composite material was measured by cyclic voltammetry. Fig. 6 shows the cyclic voltammograms in 1 M H₂SO₄ electrolyte in the range of $0 \sim 800$ mV at 50 mVs⁻¹. In order to determine a good composition ratio of the TiO₂/CNT electrode, the capacitances with different ratios of TiO₂ contents were measured in $1M H_2SO_4$. The specific capacitance C_s can be calculated from the cyclic voltammetry according to the equation (2):

$$C_{\rm s} = \left[\left(i_{\rm c} + i_{\rm a} \right) / 2 / \upsilon \right] / w \tag{2}$$

where i_c and i_a are currents at 0.0 V during the cathodic and anodic sweeps, respectively, v is the sweep rate, and w is the total weight of the active material in one electrode. The anomalous increase in the current was caused by an enhancement of the electronic and/or ionic transfer between TiO₂/CNT nanocomposite electrode and electrolyte. When TiO₂ content was small, its double layer capacitance was only slightly higher than that of pristine CNT. But with increasing the TiO₂ content ratio, the capacitance was decreased on account of improvement in diffusion resistance from impedance analysis of Fig. 7. From the result, we suggest the optimized content of TiO₂ to be 4 wt % in H₂SO₄ electrolyte. The specific capacitance of the pristine CNT and TiO₂/CNT composite electrodes are listed in Table 1. The 4 wt % TiO₂/CNT composite electrode delivered a specific capacitance of 37 Fg⁻¹ which is higher than CNT electrode by 15 Fg⁻¹.

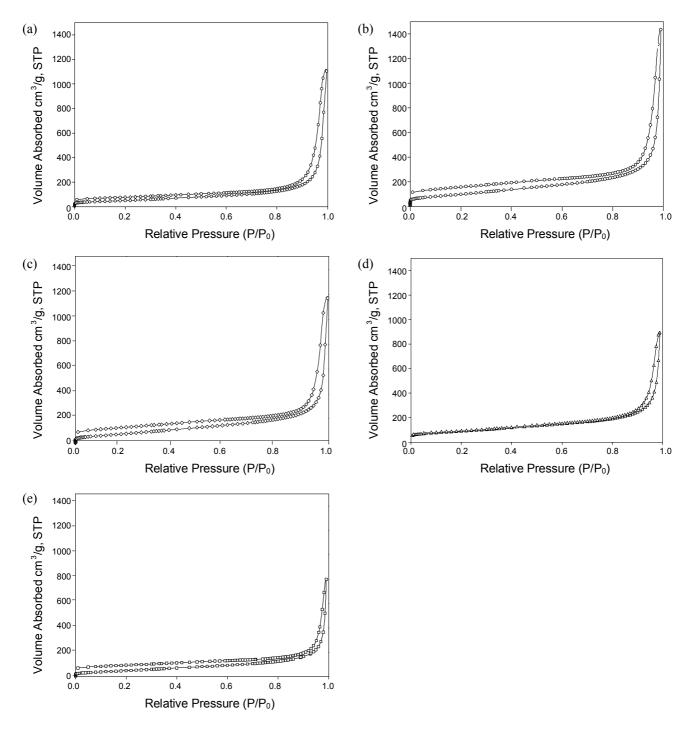


Figure 4. Nitrogen adsorption/desorption isotherms of the (a) the pristine CNT and TiO_2/CNT nanocomposites with deposited TiO_2 contents of (b) 4 wt %, (c) 6 wt %, (d) 17 wt % and (e) 32 wt %.

It has been reported²⁰ that the capacitance of an electrochemical capacitor is proportional to ion concentration and inversely proportional to the thickness of the double-layer, and the ion concentration is affected by the voltage between CNT electrodes and electrolyte. Based on above results, the increase of the capacitance in Table 1 could be attributed to the reduced polarization of CNT electrodes, which is believed to relate to the concentration of electric charges on TiO₂ surface with increasing wettability. And electric field adsorption of electrolyte on the CNT

composite electrode was significantly enhanced by TiO_2 introduction due to the increase in the charge density of ions under electric field.²¹

A more detailed estimation of the electrochemical properties of the pristine CNT and TiO₂/CNT composite electrodes can be carried out by analysis of the impedance spectra. Fig. 7 shows Nyquist plots of the pristine CNT and TiO₂/CNT composite electrodes in 1M H₂SO₄. The plot is composed of a semicircle at high frequency and a nearly vertical line at low frequency. The

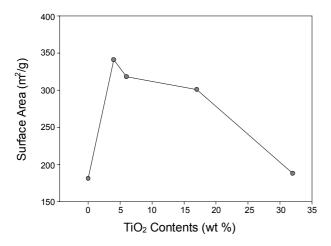


Figure 5. BET Surface area of the pristine CNT and TiO₂/CNT nanocomposites with various deposited TiO₂ contents.

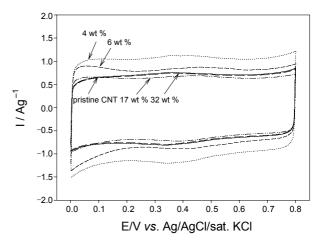


Figure 6. Cyclic voltammograms of the pristine CNT and TiO_2/CNT composite electrodes at scan rate of 50 mVs⁻¹ in 1M H₂SO₄ electrolyte.

Table 1. BET surface area, specific capacitance and discharge capacitance of the CNT and TiO₂/CNT nanocomposites

Content (TiO ₂ wt %)	BET (m^2g^{-1})	Micro pore area (m^2g^{-1})	Specific capacitance (Fg ⁻¹)	Discharge capacitance (Fg ⁻¹)
0	181	10	22	8
4	341	13	37	11
6	318	17	30	9
17	301	13	22	8
32	188	2	23	5

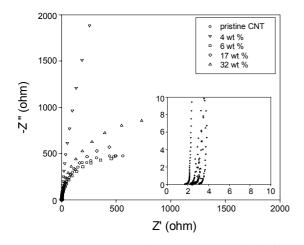


Figure 7. Nyquist plots of the pristine CNT and TiO_2/CNT nanocomposites with various deposited TiO_2 contents.

high frequency intercepts the real-axis bulk resistance of composite electrodes. The semicircle consists of electrode resistance of the pristine CNT and TiO₂/CNT composite electrodes. The bulk resistance of the capacitor built from CNT electrode was $1.7 \sim 1.8 \Omega$ at 5.6 kHz. The resistances of the CNT electrodes with TiO₂ ratio of 4 wt %, 6 wt %, 17 wt % 32 wt % showed 3.8, 2.3, 2.4 and 2.4 Ω at 7.9, 12.5, 7.9 and 5 kHz, respectively. The resistances of the electrode/electrolyte interface on TiO₂/CNT composite electrodes were slightly higher than CNT electrode. The considerable effect of the Warburg impedance on the 4 wt % TiO₂/CNT composite electrode shows a good capacitor-like behavior of near vertical slope to 10 mHz. The Warburg imped-

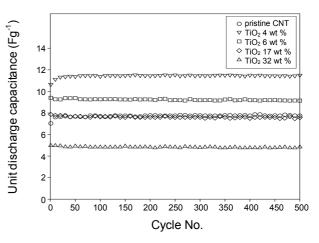


Figure 8. Cycle life performance of the pristine CNT and TiO_2/CNT electrodes at 0.5 mAcm⁻² in 1M H₂SO₄ electrolyte.

ance represents that the more the sloping line nears to the imaginary-axis, the more bulk resistance decreases. A Warburg diffusion element is incorporated in the circuit to emphasize the distributed resistance for ion diffusion in the micropores.²² The results indicate that the composite material reduced the polarization and improved the hydrophilicity of the CNT surface, resulting in the decrease of internal resistance between electrode and electrolyte and increase of the efficiency of charge and masstransfer on the electrode/electrolyte interface.⁵

This result of Nyquist plot agreed well with that from the cycle life performance. Fig. 8 shows the variation in unite discharge capacity as a function of cycle number (voltage range

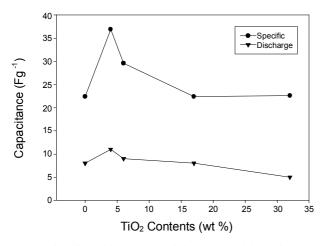


Figure 9. The effect of the contents of TiO_2 nanoparticle on the capacitance.

between 0 and 1 V, at 0.5 mAcm⁻²). The discharge capacitance (C_d) was calculated from the equation (3):

$$C_{\rm d} \left({\rm Fg}^{-1} \right) = i\Delta t \,/m\Delta V \tag{3}$$

where *i* is the current density, Δt is the discharge time, *m* is the mass of the composite material, and ΔV is the potential range of the charge–discharge cycle. Regardless of the TiO₂ contents, symmetric cells showed the similar variation trend in unit discharge capacity with cycle : all capacitances showed a small loss in the first 10 cycles, and then remained almost constant as cycling continued. The average values of unit discharge capacity for the TiO₂/CNT composite electrodes and CNT electrode are listed in Table 1. The electrodes with 4 wt % and 6 wt % deposited TiO₂ not only delivered a unite discharge capacity of 11 Fg⁻¹ and 9 Fg⁻¹, respectively, but also maintained an excellent cycling stability.

The dependence of specific capacitance on TiO_2 contents is shown in Fig. 9. The CNT composite electrode with 4 wt % of deposited TiO_2 shows higher capacitance and surface area than pristine CNT. From the result, it is indicated that increasing surface area and reducing polarization increase specific capacitance of CNT with TiO_2 nanoparticle. Table 1 summarizes the values of all the surface area and capacitance for the TiO_2/CNT composite electrodes along with the pristine CNT.

Conclusion

In the present work, CNT-based TiO₂ nanocomposites were successfully prepared by a simple method using ultrasound, and TiO₂ nanoparticles (< 20 nm) were partly deposited onto CNT. Nanocomposites electrodes with $4 \sim 6$ wt % of deposited TiO₂ presented a better BET surface area compared with the pristine CNT. From Potentiostat/Galvanostat measurements in 1M H₂SO₄ aqueous solution, the specific capacitance and electrochemical performance improved through titanium oxide composition introduction. The specific capacitance of the comHong-Il Kim et al.

posite electrode increased from 22 Fg^{-1} to 37 Fg^{-1} in comparison with the pristine CNT from cyclic voltammogram. And cycle life performance of the symmetric cell showed increase in capacitance from 8 Fg^{-1} to 11 Fg^{-1} and maintained an excellent cycling stability. We suggest the optimized content of TiO₂ to be 4 wt % and the increase in specific capacitance can be attributed to the decrease in electric polarization, caused by the deposition of TiO₂ nanoparticles on the CNT. This conclusion is supported by the Warburg impedance behavior showed near vertical slope to 10 mHz.

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