

Dielectric properties of ZrTiO₄ thin films deposited by DC magnetron reactive sputtering

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Abstract – Paraelectric ZrTiO₄ thin films were synthesized on a Si (100) substrate using DC magnetron reactive sputtering. Films deposited above 400°C exhibited crystalline characteristics. The dielectric constants (ϵ) and dielectric losses ($\tan \delta$) of as-deposited and annealed films were measured in the 1 MHz range using a Pt upper electrode and a phosphorous-doped Si bottom electrode. Preliminary data showed that as the deposition temperature increased, the dielectric losses decreased while the dielectric constants did not change significantly. Similar trends for dielectric losses were observed when the as-deposited samples were annealed at 800°C. The reduction of dielectric losses at high-deposition temperatures and post annealing correlated well with the x-ray diffraction peak widths.

I. Introduction

Recently, demand for monolithic microwave integrated circuit technologies (MMIC) is increasing with wide microwave communication applications, such as mobile phones, global positioning systems (GPS), and satellite communications. Continued miniaturization of integrated circuitry requires microwave components with improved characteristics, smaller size, and compatibility with existing circuits [1-4]. Generally, dielectric materials used for resonators and filters in microwave circuits must exhibit high dielectric constants (because the resonator size is proportional to $\sqrt{\epsilon}$), low dielectric losses ($\tan \delta = 1/Q$, where Q is the dielectric quality factor), and temperature coefficients of resonance frequency (τ_f) [5].

It has been reported that the bulk ZrTiO₄ phase has a high dielectric constant suitable for microwave devices, low dielectric losses, and good thermal stability with Sn addition [6-8]. The single phase ZrTiO₄ structure is α -PbO₂ orthorhombic where Ti and Zr ions are chemically disordered in half of the eight octahedral sites in a single unit cell above 1100°C [9-12].

Thin films of ZrTiO₄ have been synthesized by several researchers [13-16]. However, the dielectric losses of these thin films in the microwave and MHz ranges have not been investigated yet. In this paper,

the synthesis of ZrTiO₄ thin films using DC magnetron reactive sputtering, with the effects of deposition temperature and post thermal annealing on the dielectric losses ($\tan \delta$) and dielectric constants (ϵ) in the 1 MHz range, is reported.

II. Experimental Details

Thin films of ZrTiO₄ were deposited by DC magnetron reactive sputtering using 4 inch Ti and Zr metal targets on a phosphorous-doped Si (100) substrate. The deposition system was arranged in a symmetric configuration with a rotating substrate holder for composition uniformity. The base pressure of the sputtering chamber was typically 4×10^{-7} Torr. After heating to the desired deposition temperature, the targets were pre-sputtered for at least 30 minutes in the same condition as the actual deposition process. Substrates were loaded from a load lock into the main chamber to maintain low base pressure. A SiC coated graphite block was used for heating, and the substrate temperature was calibrated with a thermocouple located between the substrate and heating block. An operating pressure of 4 mTorr was maintained during deposition by a throttle valve. The Ar (99.9999%) and O₂ (99.99%) flow rate ratios were in the range of 83/17 to 50/50, as controlled by mass flow controllers. All the films were sputtered at 500 W and 700 W for Zr and Ti

targets, respectively. For electrical measurements, 200 μm diameter platinum upper electrodes were sputtered using a shadow hard mask. The phosphorous-doped Si (100) substrate had a resistivity of $6.1 \times 10^{-4} \Omega\text{-cm}$. This made it possible to measure the electric characteristics without a separate bottom electrode. Thin film samples deposited at temperatures between room temperature and 600°C were annealed at 800°C in an oxygen atmosphere for 2 hours to investigate the effects of post annealing.

The crystal structure and film texture were characterized by x-ray diffraction (XRD) using Cu K α radiation. Film compositions were measured by electron probe micro-analysis (EPMA) on several areas for each sample. The film thickness was measured with field-emission scanning electron microscopy (FE-SEM) and α -step. The typical film thickness was found to be 400 nm. The film capacitances and dielectric losses were measured using a HP4194A impedance analyzer in the 1 MHz range with a 0.04 volt r.m.s. oscillation voltage. Using the measured capacitances, the dielectric constants were calculated with film thickness, and upper electrode area.

III. Results and Discussion

The XRD patterns for the as-deposited and post-annealed thin films are shown in Figs. 1 and 2, respectively. The XRD patterns of films deposited at room temperature exhibited amorphous characteris-

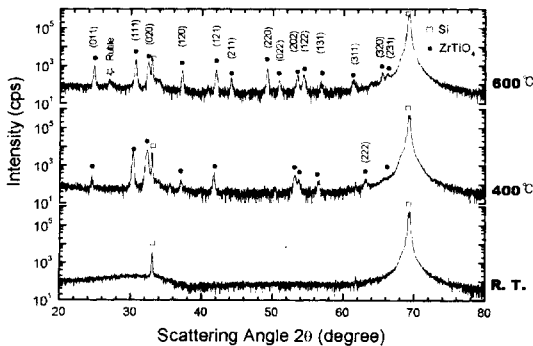


Fig. 1. X-ray diffraction pattern of the as-deposited ZrTiO₄ thin films at room temperature, 400°C, and 600°C. Diffraction peaks are indexed from the high-temperature phase ZrTiO₄.

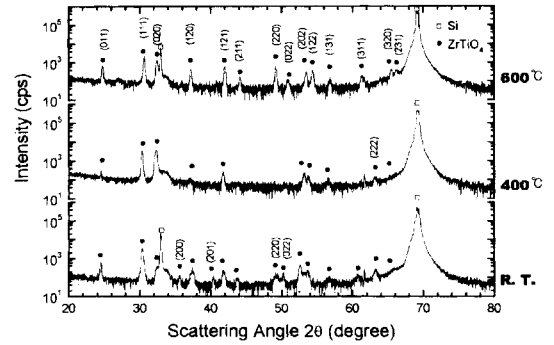


Fig. 2. X-ray diffraction pattern of ZrTiO₄ thin films annealed at 800°C in oxygen for 2 hours.

tics with only substrate peaks showing. Above 400°C, the films show polycrystalline ZrTiO₄ peaks. After annealing in an oxygen atmosphere, the films show more crystalline characteristics as shown in Fig. 2. To qualitatively estimate the effective grain size of deposited ZrTiO₄, XRD peak widths (Δk) were fitted at various stages where Δk is full width at half maximum (FWHM) for the (111) peak with the scattering vector $k = (4\pi/\lambda)\sin\theta$. Figure 3 clearly shows a reduction of the (111) peak widths after annealing at 800°C. The peak positions were found to shift slightly from a bulk value ($\sim 1\%$ larger lattice constant for deposition at 400°C). This shift was reduced at higher deposition temperatures and after the post-annealing process, indicating some structural relaxation in thin films.

Figure 4 shows the dielectric constants (ϵ) measured at 1 MHz for films at various deposition temperatures. The dielectric properties of bulk ZrTiO₄

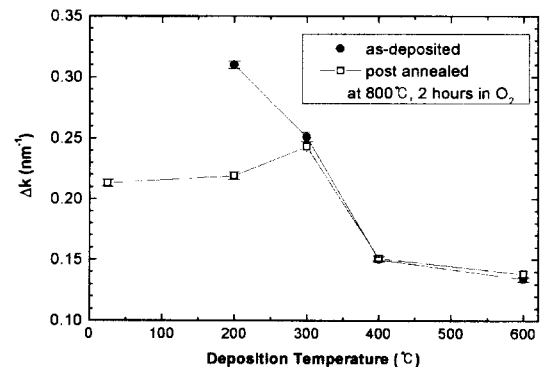


Fig. 3. ZrTiO₄ (111) peak widths (Δk), as a function of deposition temperature.

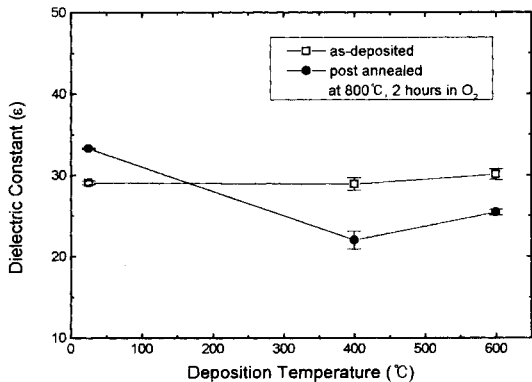


Fig. 4. Dielectric constants for as-deposited and post-annealed ZrTiO₄ thin films.

were reported to be $\epsilon = 42$, $\tan \delta = 2.13 \times 10^{-4}$ ($Q = 4700$) at 7 GHz, and $\tau_f = 58$ ppm/°C [17]. There was no systematic variation observed in the dielectric constants of the as-deposited samples as a function of deposition temperature. An increase in ϵ was expected with increasing crystallinity. It has been reported that different processing conditions may cause variations in stoichiometry and oxygen vacancies, which may result in different dielectric constants [18]. Ramakrishnan *et al.* [19] claim that the decrease in ϵ for ZrTiO₄ after annealing may be due to the formation of a low dielectric interfacial layer between films and substrates, for example SiO₂. However, these claims should be investigated further.

The change in dielectric losses ($\tan \delta$) as a function of deposition temperature is shown in Figure 5. The dielectric losses at 1 MHz were much higher than observed with bulk ZrTiO₄, but a decrease with

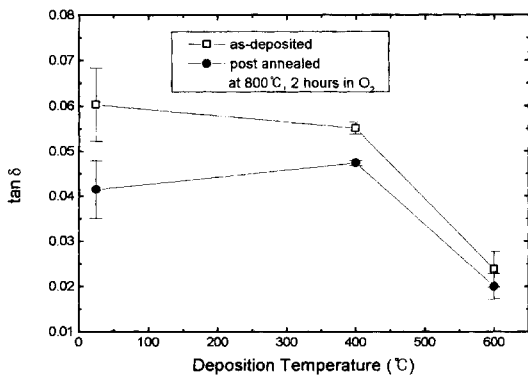


Fig. 5. Dielectric losses for as-deposited and post-annealed ZrTiO₄ thin films.

higher deposition temperature and post annealing was seen, as expected. The decrease in dielectric losses correlated with the diffraction peak widths (Figs. 3 and 5). However, it is not yet clear why ZrTiO₄ thin films have higher dielectric losses compared to the sintered bulk phase.

IV. Conclusions

The effects of deposition temperature and post annealing on the dielectric properties and crystallinity of ZrTiO₄ thin films have been investigated. Near-stoichiometric ZrTiO₄ thin films that exist in a metastable high-temperature disordered phase were successfully synthesized. As deposition temperature increased, the dielectric losses decreased while the dielectric constants were not observed to change significantly. Similar trends for dielectric losses were observed after the as-deposited samples were annealed at 800°C. However, dielectric losses in thin films were much higher than the published bulk values. Further studies are required to correlate the dielectric properties with thin film microstructures, including grain size, stoichiometry, vacancy, grain-boundary segregation, and texture.

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References

- [1] R. J. Cava, W. F. Peck, Jr., J. J. Krajewski, G. L. Roberts, B. P. Barber, H. M. OBryan and P. L. Gammel, *Appl. Phys. Lett.* **70**, 1396 (1997).
- [2] G. C. Liang, R. S. Withers, B. F. Cole and N. Newman, *IEEE Trans. Microwave Theory Tech.* **42**, 34 (1994).
- [3] D. E. Oates and A. C. Anderson, *IEEE Trans. Magn.* **28**, 867 (1991).
- [4] K. H. Young, G. V. Negrete, R. B. Hammond, A. Inam, R. Ramesh, D. L. Hart and Y. Yonezawa, *Appl. Phys. Lett.* **58**, 1789 (1991).
- [5] W. Wersing, in *Electronic Ceramics*, edited by B. C.

- H. Steele (Elsevier Science Publishers LTD, New York, 1991), p. 67.
- [6] G. Wolfram and E. Gobel, *Mater. Res. Bull.* **16**, 1455 (1981).
- [7] Y. Higuchi and H. Katube, U. S. Patent No. 4 665 041 (1987).
- [8] Y. Park and Y. Kim, U. S. Patent No. 5 561 090 (1996).
- [9] E. Newnham, *J. Am. Ceram. Soc.* **50**, 216 (1967).
- [10] R. Christoferson and P. K. Davis, *J. Am. Ceram. Soc.* **75**, 563 (1992).
- [11] F. Azough, R. Free and J. Petzelt, *J. Mater. Sci.* **28**, 2273 (1993).
- [12] A. E. McHall and R. S. Roth, *J. Am. Ceram. Soc.* **66**, C18 (1983).
- [13] D. A. Chang, P. Lin and T. Y. Tseung, *J. Appl. Phys.* **77**, 4445 (1995).
- [14] D. A. Chang, P. Lin and T. Y. Tseung, *J. Appl. Phys.* **78**, 7103 (1995).
- [15] F. J. Wu and T. Y. Tseung, *J. Am. Ceram. Soc.* **81**, 439 (1998).
- [16] O. Nakagawara, Y. Toyota, M. Kobayashi, Y. Yoshino, and Y. Katayama, *J. Appl. Phys.* **80**, 388 (1996).
- [17] F. Azough, R. Freer, C. L. Wang, G. W. Lorimer, *J. Mater. Sci.* **31**, 2539 (1996).
- [18] Y. Liang and D. Bonnell, *J. Am. Ceram. Soc.* **78**, 2633 (1995).
- [19] E. S. Ramakrishnan, K. D. Cornett, G. H. Shapiro, and W. Y. Howng, *J. Electrochem. Soc.* **145**, 358 (1998).