

사파이어 기판 위에 펄스-증착법으로 성장한 YBCO/CeO₂ 박막의 초전도성과 표면 모폴로지

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Superconductivity and Surface Morphology of YBCO/CeO₂ Thin Films on Sapphire Substrate by Pulsed Laser Deposition

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Abstract - The crystal structure and properties of YBa₂Cu₃O_{7-x}(YBCO) and CeO₂ thin films deposited on *r*-plane (1102) sapphire substrate by pulsed-laser deposition(PLD) have been investigated. *C*-axis oriented epitaxial YBCO thin films with critical temperature (T_c) of 88 K were routinely grown on (200) oriented CeO₂ buffer layers with thickness in the range between 20 to 80 nm. When the thickness of the (200)oriented CeO₂ buffer layer increases over than 80 nm, the superconducting properties of YBCO thin films on that were deteriorated. The decrease in T_c of YBCO thin films was explained by the microcrack formation in CeO₂ buffer layer. These results indicate that the thickness of the (200) oriented CeO₂ buffer layer is critical to the epitaxial YBCO thin film growth on *r*-plane(1102) sapphire substrate.

1. INTRODUCTION

Owing to the good crystalline perfection, mechanical strength, low dielectric constant and losses, *r*-plane (1012) sapphire substrate is recognized as a very promising for the deposition of epitaxial YBCO films for microwave device applications.

However, due to the large lattice mismatch between YBCO and sapphire and the diffusion of Al atoms from sapphire into YBCO during film deposition conditions, buffer layers such as SrTiO₃[1], YSZ[2], CeO₂[3] are used to overcome these problems. Among them, CeO₂ is recognized as a desired buffer layer for the epitaxial growth of superconducting YBCO thin

films on sapphire substrate[4-8]. Most previous studies centered on the epitaxial growth of CeO₂ thin films on various substrates by using different deposition technique. In addition, the relationship between the thin film properties of CeO₂ buffer layer and superconducting properties of YBCO thin film has not been investigated in detail.

In this study, we report on the growth of (200) oriented CeO₂ buffer layers and YBCO thin films on *r*-plane (1012) sapphire substrate. By optimizing the (200) oriented CeO₂ film growth, YBCO thin films with high quality *c*-axis orientation could be deposited on *r*-cut plane(1012) of sapphire substrate by pulsed laser deposition

2. EXPERIMENTAL PROCEDURE

Both the CeO₂ and YBCO thin films were deposited on *r*-plane (1102) sapphire substrate by pulsed laser deposition technique. The 248 nm output from a KrF excimer laser of 3 J/cm² was focused onto the rotating target at an incident angle of 45° and a repetition rate of 5 Hz. During the CeO₂ buffer layer deposition, substrate temperature was fixed at 750 °C and oxygen pressure was maintained at 180 mTorr. The detailed YBCO thin film growth conditions are described elsewhere[9,10]. The structure of the as-grown YBCO thin films was examined by X-ray diffractometer (XRD) with *CuKα* radiation. Surface morphologies were observed by scanning electron microscopy (SEM). Resistance-temperature(R-T) characteristics

were measured by a conventional four-probe method.

3. RESULTS AND DISCUSSION

The crystalline orientation of CeO₂ thin films was strongly affected by the deposition temperatures. Figure 1 shows the XRD patterns for the 300 nm thick CeO₂ thin films deposited at various deposition temperatures. For the case of CeO₂ thin films deposited above 750°C, only the CeO₂(200) peak was dominant. However, with decreasing the deposition temperature into 750°C, the CeO₂(111) peak was observed in addition to CeO₂(200) peak.

At 650°C deposition temperatures, the CeO₂(111) was dominant. From these results, in order to use as a buffer layer for the YBCO thin film growth on sapphire substrate, we fixed the growth temperature of CeO₂ buffer layer to 750°C.

The superconductivity of YBCO thin films with 430 nm thick was depended significantly on the thickness of the (200) oriented CeO₂ buffer layer. Influence of the (200) oriented CeO₂ buffer layer thickness on T_c of YBCO thin film is shown in Fig. 2. The T_c of the YBCO thin film directly deposited on a sapphire substrate was 63 K. When the (200) oriented CeO₂ buffer layer thickness is in the range of 20 to 80 nm, YBCO thin films with a T_c of about 88K are obtained reproducibly. For the thickness of the (200) oriented CeO₂ buffer layer increases over 80 nm, YBCO thin films showed a T_c of 83 K. From these results, it is considered that the effective thickness of CeO₂ as buffer layers for YBCO thin film growth on sapphire substrate are 20 to 80 nm.

Figure 3 shows the SEM images of YBCO thin films deposited on (200) oriented CeO₂ buffer layer with thickness in the range from 0 to 450 nm. YBCO thin films deposited directly on sapphire substrate show the surface morphology composed of large-island shape grains with a low connectivity, high density of grain boundaries and pores. As the (200) oriented CeO₂ buffer layer thickness increases from 0 to 160 nm, YBCO thin films have a smooth surface and the pore size decreased. However, when the (200) CeO₂ oriented buffer layer thickness increases near to 450 nm, YBCO thin film has a surface morphology with high density of grain boundaries, micro-size pores, and microcracks. In order to explain the

porous surface morphology of YBCO thin film deposited on (200) oriented CeO₂ buffer layers with 450 nm thickness, we examined the CeO₂ surface morphologies with various thicknesses.

As shown in Fig. 4, The (200) oriented CeO₂ buffer layers with a thickness below 80 nm have a smooth surface. However, as the thickness of the CeO₂ buffer layer increases to 160 nm, microcracks were observed. These microcracks could be seen clearly in the (200) oriented CeO₂ buffer layer with 450 nm thickness. These microcracks seem to be originated from the mismatch of lattice constants and the thermal expansion coefficients of sapphire, CeO₂ and YBCO(111). From these results, we could understand that the porous surface morphology and low T_c values of YBCO thin film deposited on (200) oriented CeO₂ buffer layers with 450 nm thickness are due to the microcrack formation in (200) oriented CeO₂ buffer layer.

4. CONCLUSIONS

We have investigated the influence of the CeO₂ buffer layer thickness on the superconducting and microstructure properties of YBCO thin films on *r*-plane (1102) sapphire substrates deposited by PLD technique. The (200) oriented CeO₂ buffer layer growth was notable for the deposition temperature above 750°C. The (200) oriented CeO₂ buffer layers deposited below 80 nm thickness showed smooth surface morphology. However, the (200) oriented CeO₂ buffer layer with 450 nm thickness showed rough surface morphology composed of rectangular shape grains with microcracks.

High quality *c*-axis oriented YBCO thin film with T_c of 88K and a smooth surface were routinely obtained on (200) oriented CeO₂ buffer layer with a thickness in the range between 10 to 80 nm. When the thickness of the (200) oriented CeO₂ buffer layer increases over than 80 nm, the superconducting properties of YBCO thin films were deteriorated. These results were considered due to the microcrack formation in YBCO and CeO₂ buffer layer films. As a results of the investigation, it was found that (200) oriented CeO₂ buffer layer deposited at 750°C with a thickness of about 40 nm is best buffer layer conditions for YBCO thin film growth on *r*-plane (1102) sapphire substrate.

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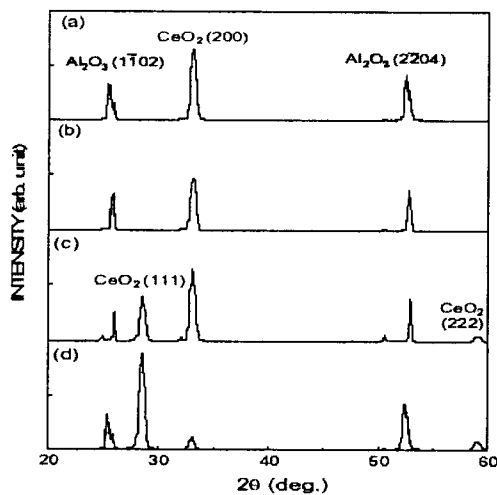


Fig. 1. X-ray diffraction patterns for the CeO₂ buffer layers deposited at various deposition temperatures of (a) 800°C, (b) 750°C, (c) 700°C, and (d) 650°C.

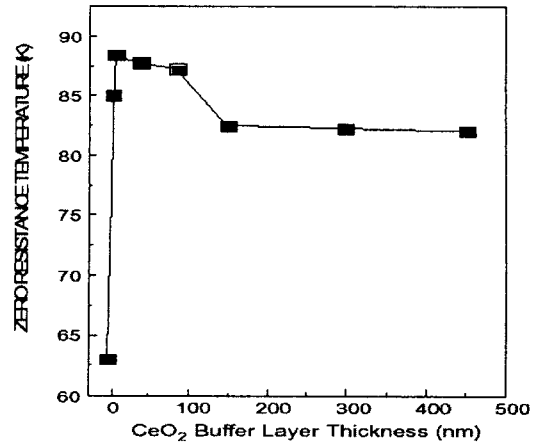


Fig. 2. Resistance vs. temperature characteristics of YBCO thin films deposited at 750°C with different as a function of the (200) oriented CeO₂ buffer layer thickness.

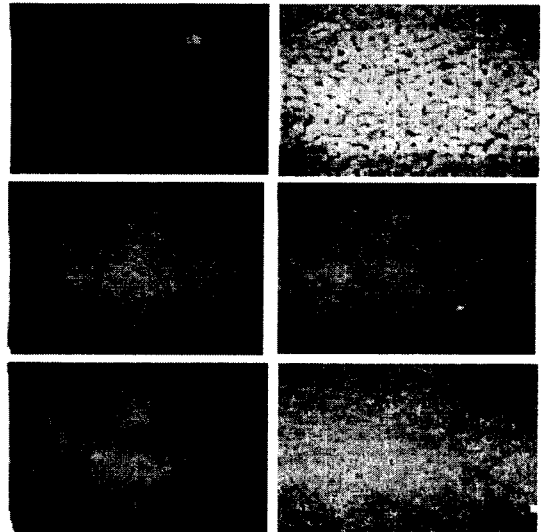


Fig. 3. Scanning electron micrographs for YBCO thin films deposited on the (200) oriented CeO₂ buffer layer of (a) 0 nm, (b) 10 nm, (c) 40 nm, (d) 80 nm, (e) 160 nm, and (f) 450 nm.

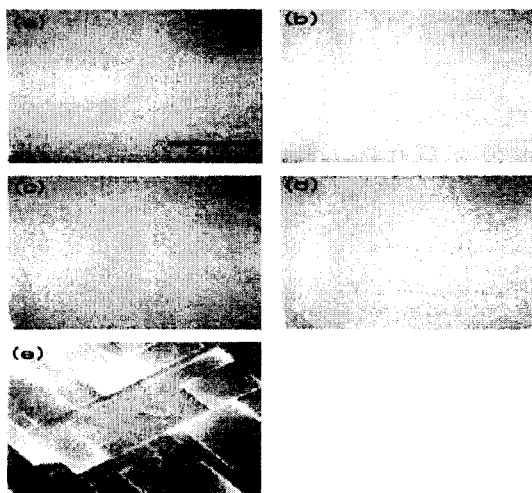


Fig. 4. Scanning electron micrographs for the (200) oriented CeO_2 buffer layers deposited at 750°C with different thickness of (a)20 nm, (b)40 nm, (c)80 nm, (d)160 nm. and (e)450 nm.