Facile Low-temperature Chemical Synthesis and Characterization of a Manganese Oxide/multi-walled Carbon Nanotube Composite for Supercapacitor Applications

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Mn₃O₄/multi-walled carbon nanotube (MWCNT) composites are prepared by chemically synthesizing Mn₃O₄ nanoparticles on a MWCNT film at room temperature. Structural and morphological characterization has been carried out using X-ray diffraction (XRD) and scanning and transmission electron microscopies (SEM and TEM). These reveal that polycrystalline Mn₃O₄ nanoparticles, with sizes of about 10–20 nm, aggregate to form larger nanoparticles (50–200 nm), and the Mn₃O₄ nanoparticles are attached inhomogeneously on MWCNTs. The electrochemical behavior of the composites is analyzed by cyclic voltammetry experiment. The Mn₃O₄/MWCNT composite exhibits a specific capacitance of 257 Fg^{−1} at a scan rate of 5 mVs^{−1}, which is about 3.5 times higher than that of the pure Mn₃O₄. Cycle-life tests show that the specific capacitance of the Mn₃O₄/MWCNT composite is stable up to 1000 cycles with about 85% capacitance retention, which is better than the pure Mn₃O₄ electrode. The improved supercapacitive performance of the Mn₃O₄/MWCNT composite electrode can be attributed to the synergistic effects of the Mn₃O₄ nanoparticles and the MWCNTs, which arises not only from the combination of pseudocapacitance from Mn₃O₄ nanoparticles and electric double layer capacitance from the MWCNTs but also from the increased surface area, pore volume and conducting property of the MWCNT network.

Key Words : Carbon nanotubes, Chemical synthesis, Composite materials, Manganese oxides, Supercapacitor

Introduction

Supercapacitors have attracted much attention because of their unique capability as an energy storing device which can overcome the drawbacks of conventional capacitors and batteries. Supercapacitors not only exhibit fast charging/discharging time, superior cycling life, and high reliability but also they can provide higher energy density than conventional capacitors and higher power density than batteries.^{1–3} Owing to the role of supercapacitors, which may bridge the gap between conventional capacitors and batteries, they have been utilized in a wide range of applications such as electric vehicles, electronic appliances, digital communication devices, and mobile phones. Carbon materials, transition metal oxides and conducting polymers have been explored as potential electrode materials used in supercapacitors.^{4–7} Among various transition metal oxides, ruthenium oxide is one of the most promising pseudocapacitor electrode materials because of its superior electrochemical response.⁸ However, despite the superior performance of ruthenium oxide, it is still limited in its industrial supercapacitor applications due to its high cost.⁹ Hence, considerable effort is being devoted to developing inexpensive metal oxide electrode materials with excellent supercapacitive performance. Among various alternative materials investigated, manganese oxides have attracted tremendous interest because

of their high theoretical specific capacitance, natural abundance, and environmentally-friendly nature.^{10,11} Especially, hausmannite (Mn₃O₄) is potentially interesting materials, because of its unique structural features combined with physicochemical properties, which are of great interest in energy fields.¹² However, only a few studies have been done with Mn₃O₄, because of its low electrical conductivity limiting the performance of Mn₃O₄ such as capacitance. An increase in mass loading and film thickness of manganese oxide further deteriorates the capacitive performance. To compensate for the poor electrical conductivity and dense morphology of the manganese oxide film, a thin layer of manganese oxide is deposited on the surface of a highly conductive and porous structure with high surface area, which can provide excellent supercapacitive performance with high mass loading of manganese oxide. The porous structure may be a metal foam, carbon fabric, graphene, or carbon nanotube film.^{13–17} Recent efforts to integrate Mn₃O₄ with porous structures have mostly focused on depositing Mn₃O₄ thin layers onto carbon nanotube (CNT) network films. The porous structures in CNT network films may offer effective pathways for electrolyte diffusion, and the highly conductive CNT network can act as a charge transportation pathway. In addition, CNTs are mechanically strong and chemically stable. Various methods have been employed for depositing Mn₃O₄ layers onto CNT films,

including successive ionic layer adsorption and reaction (SILAR),¹⁸ dip-casting,¹⁹ electrophoretic deposition²⁰ and chemical deposition.²¹ The drawbacks of these methods are that they contain time-consuming multiple steps and high-temperature processes for the synthesis of Mn_3O_4 . Therefore, more environmentally friendly, faster, and energy-efficient synthetic methods are of interest. The chemical deposition method is one of the simplest methods to deposit thin films of nanomaterials on various substrates and is a scalable technique that can be employed for large area deposition. In the chemical deposition method, the morphology and film thickness can be controlled by adjusting growth conditions such as temperature, pH, concentration, and solution composition. The morphology and growth of thin films also depend on the topographical and chemical nature of the substrate.²²⁻²⁶

In the present work, we report a simple and cost-effective chemical method to grow Mn_3O_4 nanoparticles onto multi-walled carbon nanotube (MWCNT) network films at room temperature, and the Mn_3O_4 /MWCNT composite film is further utilized as a supercapacitor electrode. The Mn_3O_4 /MWCNT composite films have been characterized by a variety of techniques, including X-ray diffraction (XRD), Brunauer-Emmett-Teller (BET) surface area measurement, field emission scanning electron microscopy (FESEM), transmission electron microscopy (TEM) and energy dispersive X-ray (EDX) analysis. The electrochemical properties of Mn_3O_4 /MWCNT composite electrodes are examined by cyclic voltammetry measurement, and the results reveal significant improvement in supercapacitive performance compared to pure Mn_3O_4 electrode.

Experimental

Analytical grade chemicals of manganese (II) chloride tetrahydrate (99.999%) and ammonium hydroxide solution (28% NH_3 in H_2O , $\geq 99.99\%$) were purchased from Sigma-Aldrich and used without additional purification. The MWCNTs with 15–20 nm diameters and 95% purity were purchased from Iljin Nanotech, Korea. Surface functionalization of the MWCNTs and their thin film formation on stainless steel (SS) substrates were carried out as described in previous reports,^{27,28} and the Mn_3O_4 nanoparticles were deposited on the MWCNT-coated SS substrates by using a chemical deposition method.²⁶ Briefly, an aqueous solution of 0.1 M manganese chloride was prepared, and the pH of the solution was adjusted with ammonia solution to 8.5, which was used as the precursor solution for the deposition of the Mn_3O_4 films. The MWCNT-coated SS substrates were immersed in this solution for 24 h at room temperature. During precipitation, heterogeneous reactions occurred, and deposition of Mn_3O_4 nanoparticles occurred on the substrate. The Mn_3O_4 /MWCNT/SS substrate was removed from the bath and washed with double distilled water, dried in air and then characterized. Similarly, the Mn_3O_4 film was deposited on an SS substrate for comparison. The weights of the samples were measured by taking the difference of the

weights of a substrate before and after film deposition. The weight percentage of CNTs in the composite is 20%, and the mass loading of the active material (Mn_3O_4 /MWCNT composite) used was 0.54 mg.

A crystallographic study was carried out using an X-ray diffractometer (Rigaku 2500) with $\text{CuK}\alpha$ radiation ($\lambda = 1.5418 \text{ \AA}$). The surface morphology of the Mn_3O_4 and Mn_3O_4 /MWCNT composite films were investigated by field emission scanning electron microscopy (FESEM, Hitachi S4800) and transmission electron microscopy (TEM, JEOL JEM-2100F) operated at 200 kV. Compositions of the films were determined by energy dispersive X-ray (EDX) analysis. The Brunauer-Emmett-Teller (BET) specific surface areas were obtained from nitrogen adsorption/desorption isotherms recorded at -196°C . The electrochemical measurements were carried out in a typical three-electrode experimental cell equipped with a working electrode, platinum counter electrode and Ag/AgCl reference electrode. All the supercapacitor characteristics were measured using 1 M aqueous Na_2SO_4 solution as an electrolyte. Cyclic voltammetry (CV) measurement was performed using a CHI 660D electrochemical workstation (CH instrument, USA) to determine the electrochemical properties. Electrochemical impedance spectroscopy (EIS) measurements were recorded at the open circuit potential in the frequency range of 0.1 Hz to 1 MHz.

Results and Discussion

Figure 1 presents XRD patterns of the Mn_3O_4 /MWCNT, Mn_3O_4 , and MWCNT films on SS substrates. An XRD pattern of the SS substrate is also included in the figure for comparison. The Mn_3O_4 /MWCNT and Mn_3O_4 films show peaks at 18.00° (101), 28.88° (112), 32.32° (103), 36.09° (211), 37.98° (004), and 59.84° (224). These peaks match well with the standard pattern of tetragonal Mn_3O_4 (JCPDS 24-0734) without any collateral peaks, indicating high purity of the Mn_3O_4 . Note that the characteristic graphitic peak of the MWCNTs at 26.22° (002) is observed in both the MWCNT and Mn_3O_4 /MWCNT film, and some peaks result from the SS substrates because the XRD analysis was performed on a thin film on the substrate.

As seen in Figure 2(a), Mn_3O_4 nanoparticles with diameters of 50 to 200 nm are deposited on SS substrate by the chemical deposition method. A high magnification image (Fig. 2(b)) further reveals a rough surface of individual nanoparticles. Figures 2(c) and 2(d) exhibit FESEM images of the Mn_3O_4 /MWCNT films. As can be seen from the figures, the Mn_3O_4 /MWCNT forms a highly porous three-dimensional network film with Mn_3O_4 nanoparticles. This porous network structure provides a large amount of surface area and facilitates ion diffusion of electrolyte, and the electrically conductive MWCNT network frame enhances the charge transfer rate, which are beneficial for supercapacitive performance. The Brunauer-Emmett-Teller (BET) specific surface area measurements show that the surface area of the Mn_3O_4 /MWCNT is $103 \text{ m}^2\text{g}^{-1}$, which is much larger than the value of $38 \text{ m}^2\text{g}^{-1}$ for pure Mn_3O_4 . In addition,

Figure 1. X-ray diffraction patterns of a stainless steel substrate, Mn_3O_4 , MWCNT, and $\text{Mn}_3\text{O}_4/\text{MWCNT}$ samples.

the total pore volume of Mn_3O_4 and $\text{Mn}_3\text{O}_4/\text{MWCNT}$ are 0.204 and $0.682 \text{ cm}^3\text{g}^{-1}$, respectively. Therefore, $\text{Mn}_3\text{O}_4/\text{MWCNT}$ is expected to exhibit superior electrochemical properties than the pure Mn_3O_4 .

TEM was used to further investigate structural and crystalline characteristics of the $\text{Mn}_3\text{O}_4/\text{MWCNT}$ films. Figure 3 exhibits TEM images of the $\text{Mn}_3\text{O}_4/\text{MWCNT}$ films. As seen in Figure 3(a), Mn_3O_4 nanoparticles, with sizes of about 10–20 nm, aggregate and form larger nanoparticles with sizes about 50–200 nm. The Mn_3O_4 nanoparticles are attached

Figure 2. FESEM images of Mn_3O_4 (a and b) and $\text{Mn}_3\text{O}_4/\text{MWCNT}$ (c and d) on stainless steel substrates.

Figure 3. TEM images of $\text{Mn}_3\text{O}_4/\text{MWCNT}$ at low (a) and high (b) magnifications. The inset shows the selected area electron diffraction (SAED) pattern.

inhomogeneously on the MWCNTs. High resolution TEM (Fig. 3(b)) shows the nanoparticles have aligned lattice fringes and measured interfringe distances of 0.493, 0.309 and 0.249 nm, corresponding to (101), (112) and (211) d-spacing of tetragonal Mn_3O_4 atomic plane orientation, respectively. For further analysis of crystalline structure of manganese oxide nanoparticles, the selected area electron diffraction (SAED) was conducted, and the SAED pattern is shown in Figure 3(b) (inset). The diffuse, spotty bright electron diffraction rings indicate a nanocrystalline nature of the Mn_3O_4 particles. Most of the rings are contributed by the characteristic crystal planes of Mn_3O_4 , which is consistent with the XRD results. The composition of the $\text{Mn}_3\text{O}_4/\text{MWCNT}$ sample was determined by EDX (data not shown here). Only carbon, oxygen, manganese, and copper (from the Cu grid) peaks were detected, suggesting the formation of $\text{Mn}_3\text{O}_4/\text{MWCNT}$ composite.

Cyclic voltammetry (CV) was employed to investigate electrochemical properties of the Mn_3O_4 and $\text{Mn}_3\text{O}_4/\text{MWCNT}$ electrodes. Figure 4(a) shows the CV curves of Mn_3O_4 and $\text{Mn}_3\text{O}_4/\text{MWCNT}$ electrodes, which were measured at a scan rate of 50 mVs^{-1} in 1 M Na_2SO_4 electrolyte within a potential window ranging from -0.1 to $+0.9 \text{ V}$ (vs. Ag/AgCl). As seen in the figure, the pure Mn_3O_4 and $\text{Mn}_3\text{O}_4/\text{MWCNT}$ electrodes both show the rectangular-like CV curves with no obvious redox peaks, which is the typical capacitive behavior of a manganese oxide electrode. The $\text{Mn}_3\text{O}_4/\text{MWCNT}$ electrode exhibits larger capacitive current than pure Mn_3O_4 , and the area of the CV curve for the $\text{Mn}_3\text{O}_4/\text{MWCNT}$ is larger than that of pure Mn_3O_4 , indicating better capacitive performance of the $\text{Mn}_3\text{O}_4/\text{MWCNT}$ compared to the pure Mn_3O_4 . The area under the CV curve can be used to estimate the specific capacitance of the electrode. The specific capacitance can be calculated from the CV curves using following equation:

$$C_{\text{sp}} = \frac{1}{ms(V_f - V_i)} \int_{V_i}^{V_f} I(V) dV$$

where C_{sp} is the specific capacitance (Fg^{-1}), m is the mass of the active electrode material (g), s is the potential scan rate (mVs^{-1}), V_f and V_i are the integration limits of the voltammetric curve (V), and $I(V)$ denotes the response current density (Acm^{-2}). From the CV curves, the specific capacitances

Figure 4. (a) Cyclic voltammograms of Mn_3O_4 and $\text{Mn}_3\text{O}_4/\text{MWCNT}$ electrodes at a scan rate of 50 mVs^{-1} . (b) Cyclic voltammograms of $\text{Mn}_3\text{O}_4/\text{MWCNT}$ electrode at different scan rates. (c) Specific capacitances of $\text{Mn}_3\text{O}_4/\text{MWCNT}$ electrode at different scan rates from -0.1 to $+0.9 \text{ V}$ in $1 \text{ M Na}_2\text{SO}_4$.

of Mn_3O_4 and $\text{Mn}_3\text{O}_4/\text{MWCNT}$ electrodes are 34.5 and 109.0 Fg^{-1} , respectively at a scan rate of 50 mVs^{-1} . The enhancement of the specific capacitance of the $\text{Mn}_3\text{O}_4/\text{MWCNT}$ electrode can be attributed to the higher surface area, conducting improvements by MWCNTs, and the combination of pseudocapacitive and EDLC properties, respectively, from the Mn_3O_4 nanoparticles and MWCNTs.

Figure 4(b) shows cyclic voltammograms of a $\text{Mn}_3\text{O}_4/$

Figure 5. (a) Nyquist plots and (b) cycle stability test for Mn_3O_4 and $\text{Mn}_3\text{O}_4/\text{MWCNT}$ electrodes.

MWCNT composite electrode at various scan rates. As can be seen in the figure, the shape of the CV curve does not significantly change with an increase in scan rate from 5 to 90 mVs^{-1} , further demonstrating excellent capacitive characteristics and fast response of the electrode. Figure 4(c) shows variations in the specific capacitance value of $\text{Mn}_3\text{O}_4/\text{MWCNT}$ composite as a function of scan rate. The specific capacitances of $\text{Mn}_3\text{O}_4/\text{MWCNT}$ composite electrodes are $257, 236, 181, 142, 109, 97, 93, 86,$ and 68 Fg^{-1} at $5, 10, 20, 30, 50, 70, 90, 150,$ and 200 mVs^{-1} , respectively. As shown in the figure, the specific capacitance decreases as the scan rate increases, indicating that fewer active sites are utilized for charge storage at higher scan rates. At high scan rates, the accessibility of electrolyte ions to the inner region of the porous electrode materials can be limited due to the slow diffusion of the ions within the pores of the electrode material. Therefore only the outer surface of the electrode can be utilized for the charging process, resulting in low specific capacitance. At a very slow scan rate, however, most of the active regions can be involved in the charging process,

resulting in high specific capacitance.

Figure 5(a) displays the Nyquist plots of the Mn_3O_4 and $\text{Mn}_3\text{O}_4/\text{MWCNT}$ electrode in the frequency range of 0.1 Hz–1 MHz at the open circuit potential. Both Mn_3O_4 and $\text{Mn}_3\text{O}_4/\text{MWCNT}$ electrodes similarly show an arc in the high frequency region and an inclined line in low frequency regions. Note that the charge transfer resistance (R_{ct}) at the electrode/electrolyte interface can be estimated from the diameter of the arc. A distinct difference between the two curves is that R_{ct} of $\text{Mn}_3\text{O}_4/\text{MWCNT}$ is smaller than that of Mn_3O_4 , indicating that $\text{Mn}_3\text{O}_4/\text{MWCNT}$ composite electrode has lower charge transfer resistance than pure Mn_3O_4 electrode. This result can be contributed to the enhanced electrical conductivity and reduced diffusion length of electrolyte ions in $\text{Mn}_3\text{O}_4/\text{MWCNT}$ composite electrode.

The cycling performance is also important for electrochemical supercapacitor application. Figure 5(b) demonstrates the stability characteristics of Mn_3O_4 and $\text{Mn}_3\text{O}_4/\text{MWCNT}$ electrodes as a function of cycle number obtained from the cyclic voltammetric study. It can be noticed that the specific capacitance gradually increases up to 250 cycles, which may be due to the activation effect of electrochemical cycling.^{29,30} After 1,000 cycles, $\text{Mn}_3\text{O}_4/\text{MWCNT}$ electrode can retain 85% of the initial capacity, but pure Mn_3O_4 retains only 70%. This enhancement in cycling stability can be attributed to porous structure of $\text{Mn}_3\text{O}_4/\text{MWCNT}$ in which the Mn_3O_4 nanoparticles are dispersed in the 3D MWCNTs network. Such structure can provide more space in the electrode and greater accessibility of the electrolyte ions to the available reactive sites, reducing degradation of the electrochemical performance in cyclic redox reaction.

Conclusion

A simple and cost-effective approach has been developed to fabricate $\text{Mn}_3\text{O}_4/\text{MWCNT}$ nanocomposites at room temperature. Crystalline Mn_3O_4 nanoparticles with 50–200 nm in size are distributed in the MWCNT network. The resulting $\text{Mn}_3\text{O}_4/\text{MWCNT}$ composite electrode achieves a large specific capacitance of 257 Fg^{-1} at a scan rate of 5 mVsec^{-1} , which is 3.5 times greater than that of a pure Mn_3O_4 electrode. In the $\text{Mn}_3\text{O}_4/\text{MWCNT}$ composite, the MWCNT 3D network can not only provide electrical and ionic conducting channels in an electrolyte but also high surface area for the growth of Mn_3O_4 nanoparticles. Accordingly, the $\text{Mn}_3\text{O}_4/\text{MWCNT}$ composite electrode can exhibit higher specific capacitance, lower charge transfer resistance and better cycling stability compared with the pure Mn_3O_4 electrode. These improvements result from the highly accessible specific surface area, reduced diffusion length of ions, and enhanced electrical conductivity. This work demonstrates a design for a high performance supercapacitor electrode material by synthesizing metal oxide/CNT nanocomposites by a facile and low-temperature preparation process.

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References

- Simon, P.; Gogotsi, Y. *Nature Mater.* **2008**, *7*, 845.
- Conway, B. E. *Electrochemical Supercapacitors, Scientific Fundamentals and Technological Applications*; Kluwer Academic/Plenum Publishers: New York, USA, 1997; p 15.
- Jayalakshmi, M.; Balasubramanian, K. *Int. J. Electrochem. Sci.* **2008**, *3*, 1196.
- Pan, H.; Li, J.; Feng, Y. P. *Nanoscale Res. Lett.* **2010**, *5*, 654.
- Zhang, L. L.; Xhao, X. S. *Chem. Soc. Rev.* **2009**, *38*, 2520.
- Wang, G.; Zhang, L.; Zhang, J. *Chem. Soc. Rev.* **2012**, *41*, 797.
- Snook, G. A.; Kao, P.; Best, A. S. *J. Power Sources* **2011**, *196*, 1.
- Hu, C. C.; Chen, W. C.; Chang, K. H. *J. Electrochem. Soc.* **2004**, *151*, A281.
- Wu, Z. S.; Wang, D. W.; Ren, W.; Zhao, J.; Zhou, G.; Li, F.; Cheng, H. M. *Adv. Funct. Mater.* **2010**, *20*, 3595.
- Wei, W.; Cui, X.; Chen, W.; Ivey, D. G. *Chem. Soc. Rev.* **2011**, *40*, 1697.
- Zhang, S. W.; Chen, G. Z. *Energy Mater.* **2008**, *3*, 186.
- Wang, D.; Yuqi, Y.; Wang, Q.; Wang, T. *Eur. J. Chem.* **2012**, *4*, 628.
- Lee, J. W.; Hall, A. S.; Kim, J.-D.; Mallouk, T. E. *Chem. Mater.* **2012**, *24*, 1158.
- Li, D.; Meng, F.; Yan, X.; Yang, L.; Heng, H.; Zhu, Y. *Nanoscale Res. Lett.* **2013**, *8*, 535.
- Li, Z.; Lin, N.; Wang, X.; Wang, C.; Qi, Y.; Yin, L. *J. Mater. Chem.* **2012**, *22*, 16640.
- Jiang, L.; Fan, Z. *Nanoscale* **2012**, *6*, 1922.
- Wang, T.; Peng, Z.; Wang, Y.; Tong, J.; Zheng, G. *Sci. Rep.* **2013**, *3*, 2693.
- Dubal, D. P.; Holze, R. *New J. Chem.* **2013**, *37*, 403.
- Cui, X.; Hu, F.; Wei, W.; Chen, W. *Carbon* **2011**, *49*, 1225.
- An, G.; Yu, P.; Xiao, M.; Liu, Z.; Miao, Z.; Ding, K.; Mao, L. *Nanotechnology* **2008**, *19*, 275709.
- Chen, C. Y.; Chien, T. C.; Lin, C. K.; Wang, S. C. *Diam. Relat. Mater.* **2009**, *18*, 482.
- Dubal, D. P.; Dhawale, D. S.; Salunkhe, R. R.; Fulari, V. J.; Lokhande, C. D. *J. Alloys Compd.* **2010**, *497*, 166.
- Dubal, D. P.; Dhawale, D. S.; Salunkhe, R. R.; Pawar, S. M.; Fulari, V. J.; Lokhande, C. D. *J. Alloys Compd.* **2009**, *484*, 218.
- Salunkhe, R. R.; Jang, K.; Yu, H.; Yu, S.; Ganesh, T.; Han, S. H.; Ahn, H. *J. Alloys Compd.* **2011**, *509*, 6677.
- Salunkhe, R. R.; Jang, K.; Lee, S. W.; Ahn, H. *RSC Advances* **2012**, *2*, 3190.
- Salunkhe, R. R.; Jang, K.; Lee, S. W.; Yu, S.; Ahn, H. *J. Mater. Chem.* **2012**, *22*, 21630.
- Osorio, A. G.; Silveria, I. C. L.; Bueno, V. L.; Bergmann, C. P. *Appl. Surf. Sci.* **2008**, *255*, 2485.
- Xu, H. Y.; Xu, S. L.; Wang, H.; Yan, H. *J. Electrochem. Soc.* **2005**, *152*, C803.
- Tianyu, W.; Zheng, P.; Yuhang, W.; Jing, T.; Gengfeng, Z. *Sci. Rep.* **2013**, *3*, 2693.
- Xiong, Z.; Xianzhong, S.; Haitao, Z.; Dacheng, Z.; Yanwei, M. *Electrochim. Acta* **2013**, *87*, 637.